

1 An optimization-based approach to extract faceted  
2 crystal shapes from stereoscopic images

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11 **Abstract**

The size and shape of particles crucially influences the characteristics of solid products. Until recently these quantities were evaluated using light microscopy. However, extracting the three-dimensional shape of a faceted crystal from a single image is a formidable computer vision challenge. In this work we combine stereoscopic imaging devices (e.g., commercial stereoscopic microscopes or the stereoscopic flow through cell that continuously draws samples from a crystallizer (Schorsch et al., 2014)) with a model-based approach in which parametric polytopes are used to describe faceted crystals (Hours et al., 2014). In the shape reconstruction algorithm these parametric polytopes are scaled and rotated until their projections closely match the measured stereoscopic images, which is formulated as a nonlinear optimization problem. The proposed approach is assessed using simulated images and experimental data. We also assess in which cases the proposed approach does or does not provide advantages over concepts using generic particle shapes (Schorsch et al., 2012).

12 *Keywords:* Crystallization, Particle shape, Particulate processes, Stereoscopic  
13 imaging, Faceted crystals, Nonlinear Optimization.

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## 14 **1. Introduction**

15 The particle size and shape distribution (PSSD) that results from a crys-  
16 tallization process influences solid bulk properties that are key factors during  
17 downstream processing such as filtration and drying. In the pharmaceutical  
18 industry knowledge of the PSSD of an active pharmaceutical ingredient is es-  
19 sential as the API's dissolution behavior and bioavailability critically depend  
20 on it (Variankaval et al., 2008; Lovette et al., 2008). The shape of faceted  
21 crystals, i.e., their morphology, is the result of the interplay of growth rates of  
22 individual crystal facets. Data to estimate crystal size and shape from exper-  
23 iments, either for single crystals or for entire ensembles of particles, can be  
24 obtained through imaging systems. Traditionally, crystal growth rates of indi-  
25 vidual facets are monitored using microscopic observations of stationary single  
26 crystals (e.g., Davey et al. (1986); Kitamura and Ishizu (2000)).

27 Measurement devices that acquire images of a large number of crystals that  
28 grow in a stirred tank crystallizer (and therefore in conditions that are closer  
29 to an industrial manufacturing process) have recently been developed by dif-  
30 ferent groups and have been used to demonstrate that optical microscopy or  
31 video techniques coupled with image analysis can be used to obtain useful crys-  
32 tal shape information. In situ imaging techniques that acquire data directly  
33 from the particle suspension in the crystallizer have been presented for example  
34 by Roberts and co-workers, who were able to measure two dimensional growth  
35 rates (Wang et al., 2007) and to distinguish between different crystal shapes  
36 by applying Fourier analysis (Calderon De Anda et al., 2005). Rawlings and  
37 coworkers have used in situ imaging combined with model-based object recogni-  
38 tion to characterize particles in suspension (Patience and Rawlings, 2001; Larsen  
39 et al., 2007). These initial studies motivated a substantial amount of work that  
40 has been performed to resolve image segmentation problems, i.e., the problem

41 of overlapping particles (Larsen et al., 2006; Ahmad et al., 2012; Larsen and  
42 Rawlings, 2009), which is an important consideration because the number and  
43 size of particles during a batch crystallization process usually increases sub-  
44 stantially. To circumvent this problem we have recently reported a different  
45 approach using an external flow through channel that allowed adjusting the  
46 suspension density by diluting with saturated solution drawn from the crystal-  
47 lizer (Eggers et al., 2008a). Moreover, the transparent flow through channel  
48 avoided optical distortion effects and depth of field issues that are commonly  
49 observed with in situ imaging techniques. Using this setup the evolution of  
50 two-dimensional particle size distributions during the cooling crystallization of  
51 paracetamol (Kempkes et al., 2010b) and of ascorbic acid (Eggers et al., 2008b)  
52 was successfully monitored.

53 However, all the above mentioned imaging techniques share a common issue,  
54 i.e., they suffer from the orientation dependence that occurs when a complex  
55 faceted crystal is projected on a single imaging plane. This issue can be over-  
56 come by either using a mathematical model that allows to reconstruct a PSSD  
57 from a large set of measured projections by assuming that the particles are  
58 randomly oriented (Kempkes et al., 2008; Eggers et al., 2008a) or by obtaining  
59 stereoscopic information, i.e., photographing the same particle from two differ-  
60 ent coplanar directions, therefore enabling the calculation of particle size and  
61 shape on a one to one basis. A setup based on external cameras taking pictures  
62 of suspended crystals in a reactor has been proposed by Wang et al. (2008),  
63 but an experimental implementation has – to the best of our knowledge – yet  
64 to be demonstrated in the open literature. Instead, we have presented an im-  
65 proved flow through cell in which particles are photographed from orthogonal  
66 directions using two mirrors, thus yielding a quantitative 3D particle size distri-  
67 bution (Kempkes et al., 2010a). In a subsequent work, this cell was connected

68 to a sampling and dilution loop to allow for continuous sampling and real time  
69 measurements of PSSDs using improved image analysis routines and particle  
70 size calculation protocols (Schorsch et al., 2012). The setup was then further  
71 improved in terms of fluid dynamics, image analysis software, and by replac-  
72 ing the mirrors with a second camera yielding higher quality images (Schorsch  
73 et al., 2014). Alternatively, if one is only interested in the growth kinetics, where  
74 secondary effects (agglomeration and breakage) are not an issue, stereoscopic  
75 images of still particles in solution could be obtained, e.g., by using a commercial  
76 stereo microscope (with a typical angle of 12–14° between coplanar observation  
77 directions) or a customized microscope installation with a larger angle.

78 Until recently images obtained via the above-mentioned techniques were in-  
79 terpreted by either using generic particle classes (spheres, cuboids, cylinders,  
80 etc.) (Wang et al., 2007; Schorsch et al., 2012) or by assuming that the mor-  
81 phology (i.e., which facets are exhibited on a crystal) of the crystals is known  
82 and does not change during an experiment (Calderon De Anda et al., 2005;  
83 Larsen et al., 2007), so that only the relative importance of the facets changes.  
84 However, the morphology of a crystal evolves during its growth before a “steady  
85 state growth morphology” is obtained (Zhang et al., 2006), which requires more  
86 flexible approaches to extract size information. Such approaches, in which crys-  
87 tals are described as parametric polytopes, have been presented by Borchert and  
88 Sundmacher (2012) and Hours et al. (2014) for monoscopic and stereoscopic se-  
89 tups, respectively.

90 In this work we follow the approach proposed by Hours et al. (2014) in order  
91 to reconstruct faceted crystals from two projections of the same particle. The  
92 approach presented in that paper is one that states the computer vision problem  
93 as an optimization problem in which the configuration of the polytope (i.e.,  
94 which facets are present), its scaling and its orientation are optimized such that

95 the difference between projections of the polytope and the measured stereoscopic  
96 images is minimized. In Hours et al. (2014) we presented some case studies  
97 using this approach. In the present study however, we investigate the approach  
98 more extensively for a wide variety of particle morphologies and for different  
99 crystal systems, using both simulated and experimentally obtained images. We  
100 further compare this approach in detail to our previous approach (Schorsch  
101 et al., 2014) using generic particles and identify advantages and disadvantages  
102 of both approaches in the different cases. In order to investigate if the proposed  
103 approach is applicable to commercial hot stage microscopy setups, we investigate  
104 the importance of the stereo angle, i.e., the angle between the two coplanar  
105 observation directions.

106 The remainder of this manuscript is structured as follows: First, we briefly  
107 describe the particle models used in this work (Section 2) and the data acqui-  
108 sition and image analysis procedures (Section 3); then we explain the recon-  
109 struction protocol for the above-mentioned parametric polytopes in Section 4.  
110 Finally, the applicability of the method is critically assessed for different stereo  
111 angles, different crystal systems and thus different particle morphologies in Sec-  
112 tion 5.

113 **2. Particle models**

114 Faceted crystals can be represented by convex polytopes that can be de-  
 115 scribed requiring only the crystal structure (unit cell parameters and symmetry)  
 116 and a list of  $m$  experimentally observed facets. Normal vectors  $\mathbf{a}_i$  are calculated  
 117 for every facet  $i$  based on its Miller index. Due to the symmetry of the under-  
 118 lying crystal structure some of the  $m$  facets are chemically equivalent, so that  
 119 they can be grouped into  $p$  independent facet groups. This set of normal vectors  
 120 and constraints can be used to define an inequality which separates the space  
 121 into two domains. The first is a closed body representing the crystal, which is  
 122 bounded by planes defined by the normal vectors  $\mathbf{a}_i$  and their normal distance to  
 123 the origin; while the second domain is the remaining space around the particle.  
 124 Therefore, according to this definition a crystal can be expressed as a rotated  
 125 and scaled polytope  $C$  in half space notation:

$$C(\mathbf{R}, \mathbf{t}) = \{x \in \mathbb{R}^3 \mid \mathbf{A}\mathbf{R}^T x \leq \mathbf{M}\mathbf{t}\} \quad (1)$$

126 where  $\mathbf{t} \in \mathbb{R}^p$  is a scaling parameter representing the crystal size for all  $p$  inde-  
 127 pendent growth directions as the distance of the bounding plane to the origin,  
 128  $\mathbf{R}$  is a rotation matrix,  $\mathbf{A} \in \mathbb{R}^{m \times 3}$  is the matrix that contains the normal vectors  
 129  $\mathbf{a}_i$  and  $\mathbf{M} \in \mathbb{R}^{m \times p}$  is a matrix that groups the facets into  $p$  independent facet  
 130 ensembles. Note that the shape of the polytope does not change regardless of  
 131 the rotation  $\mathbf{R}$ , however, the polytope's projections on the imaging planes are  
 132 affected and the rotation has therefore to be considered in the optimization  
 133 problem formulated in Section 4. The basis for defining matrices  $\mathbf{A}$  and  $\mathbf{M}$  is  
 134 given by symmetry of a given crystal system. Even after defining the crystal  
 135 structure (cf. Section 4) there are – in principle – an infinite number of possible  
 136 facets that could be exposed for a given crystal form. However, the list of crys-  
 137 tal facets that should be considered can be narrowed down by only considering

138 the most stable ones (typically the ones that provide opportunities for strong  
139 interactions, e.g., hydrogen bonds, with the solvent molecules) as they are most  
140 likely to be present under moderate experimental conditions (see, e.g., Lovette  
141 et al. (2008) and Kuvadia and Doherty (2011) for an in depth discussion). In  
142 Table 1 we report the structures of the crystal systems considered in this paper  
143 where the first reference given in each row reports the crystal structure and the  
144 second reference gives a list of stable facets. The matrices  $A$  and  $M$  can then be  
145 reported as in Tables 2 to 7 for the crystal systems investigated in this work.  
146 An example of a crystal polytope can be found in Figure 1 where a model of an  
147 acetaminophen crystal is shown for which the facet families  $\{201\}$ ,  $\{110\}$  and  
148  $\{001\}$  are shown in blue, red and yellow, respectively. One can easily generate  
149 two projections,  $P_1(\mathbf{R}, \mathbf{t})$  and  $P_2(\mathbf{R}, \mathbf{t})$ , of the polytope  $C$  as it is seen from the  
150 previously defined coplanar observation directions. The two projections for the  
151 acetaminophen crystal are also reported in Figure 1.

152 Depending on the scaling vector  $\mathbf{t}$  not all of the  $m$  facets might be visible on  
153 the crystal, so that domains of  $\mathbf{t}$  with a common set of constituting facets and  
154 edges can be identified, which can be accomplished using the analytical expres-  
155 sions reported by Borchert and Sundmacher (2013) and Singh and Ramkrishna  
156 (2013) or numerically as shown in Hours et al. (2014). An example of such a  
157 morphology map is given in Figure 2 for the case of acetaminophen. Differently  
158 shaded regions represent domains with common facets being present, while black  
159 lines separate regions with common edges within such a facet domain. One can  
160 thus identify five different crystal morphologies, for which exemplary polytopes  
161 are shown in Figure 3.

162 In order to compare the results obtained with the polytope representation  
163 to the particle sizing technique as demonstrated in previous works, three shape  
164 classes, that had been introduced in Schorsch et al. (2012, 2014), will be used as

165 well, i.e., a sphere with its diameter as its only characteristic length, a cylinder  
166 with diameter and length as its two characteristic lengths, and a cuboid with  
167 length, width, and depth as its three characteristic lengths.

### 168 3. Image acquisition

169 In order to assess the particle sizing techniques, two different data sources  
170 are utilized. First, artificially generated images containing projections of faceted  
171 crystals are analyzed. These images have the advantage that all aspects of the  
172 polytopes in these images, i.e., their orientation  $\mathbf{R}$  and size vector  $\mathbf{t}$  are known  
173 and can be used to compare the analysis result to the input. One may also  
174 freely choose the angle between the coplanar observation directions. Second,  
175 images of crystals from five different organic substances are photographed and  
176 investigated. In this case, a different error metric (see Section 5.1) is necessary  
177 as neither the rotation nor the size vector are known a priori.

#### 178 3.1. Generation of artificial images

179 Simulated projections of crystals are generated from the polytope model  
180 of the crystal using the *Multi-Parametric Toolbox* (MPT 3.0 (Kvasnica et al.,  
181 2012)) available for Matlab (MATLAB, 2010). The projections are then con-  
182 verted to images with the same resolution as our measurement setup using the  
183 `poly2mask` function available in the Image Processing Toolbox of Matlab.

#### 184 3.2. Stereoscopic imaging device

185 Crystals in suspension are photographed from two orthogonal coplanar di-  
186 rections using our custom built measurement device which is described and  
187 characterized in detail in Schorsch et al. (2014). In short, two cameras take  
188 pictures of crystals in suspension in a flow through cell made of sapphire glass.  
189 The crystals are conveyed to this cell and fed back into a crystallizer using a  
190 sampling loop. The projections drawn into the  $(x_1, x_3)$  and  $(x_2, x_3)$  planes in  
191 Figure 1 represent the information captured by the two cameras as a (simulated)  
192 example of a crystal as it passes through the measurement cell.

193 *3.3. Image Analysis*

194 The image analysis algorithm is the same for the artificially generated images  
195 of crystals as well as the actual photographs. It is based on `OpenCV` (Bradski,  
196 2000). The details of this procedure are described in (Schorsch et al., 2012,  
197 2014). After thresholding, contour extraction, and comparing both projections  
198 with respect to their  $x_3$  position a matching pair of projections is obtained,  
199 whose sets of boundary points  $D_1$  and  $D_2$  are then extracted:

$$D_1 = \{d_{1,i}\}_{i=1}^{N_1} \quad \text{and} \quad D_2 = \{d_{2,i}\}_{i=1}^{N_2} \quad (2)$$

200 where  $D_1$  is a dataset of coordinates of  $N_1$  boundary points of  $P_1$ , and  $D_2$  is a  
201 data set of coordinates of  $N_2$  boundary points of  $P_2$ , where we have used the  
202 shorter notation  $P_i$  for  $P_i(\mathbf{R}, \mathbf{t})$ , as always in the following. For further anal-  
203 ysis, the convex hulls,  $H_1$  and  $H_2$ , of these two data sets are calculated via a  
204 Graham scan (Graham, 1972). In order to reduce the number of data points  
205 and to remove outliers we preprocess the data by employing a Douglas-Peucker  
206 line simplification (Douglas and Peucker, 1973). The idea is to identify good  
207 candidates for the vertices of the projection. An example for an experimental  
208 image of an Ibuprofen crystal is presented in Figure 4. As one can see, this  
209 simplification introduces some discrepancy between the data points considered  
210 in the optimization algorithm and the actual projection. However, for most  
211 projections the simplification leads to good reconstruction results. The opti-  
212 mization algorithm, which will be introduced in the next section, operates on  
213 centered and normalized convex hulls, so that  $d_{j,i} \in (-1, 1)$ . In order to retrieve  
214 the actual size of the particle after the reconstruction is performed, we introduce  
215 the normalization constant  $s$ , which includes the camera magnification and the  
216 normalization information.

## 217 4. Optimization-based Polytope Reconstruction

218 The crystal shape estimation problem can be formulated as a polytope re-  
219 construction problem from a pair of projections (Hours et al., 2014). More  
220 precisely, the goal is to provide an estimate of the size vector  $\mathbf{t}$  from two sets of  
221 boundary points on projection planes, i.e., from  $D_1$  and  $D_2$  defined in Eq. (2).  
222 As the projections of a polytope depend strongly on its orientation a good esti-  
223 mate of its size vector  $\mathbf{t}$  relies on a good estimate of its orientation  $R$ , so that  $\mathbf{t}$   
224 and  $R$  need to be estimated simultaneously. Moreover, a challenging aspect of  
225 the problem is that the solutions are not unique, i.e., more than one pair of  $\mathbf{t}$   
226 and  $R$  can produce the same datasets  $D_1$  and  $D_2$ .

227 For estimating both the crystal shape and its orientation, we propose an  
228 approach based on minimizing the re-projection error between the rotated and  
229 scaled polytope  $C$  and the two sets of data points  $D_1$  and  $D_2$ .

### 230 4.1. Definition of the optimization objective

231 The first key ingredient of our method is an appropriate metric to measure  
232 the discrepancy between data and scaled model. The metric should be appropri-  
233 ate for optimization purposes and should measure the shape resemblance in a  
234 non-ambiguous way, which means that a small value of the optimization objec-  
235 tive should correspond to a good fit between the projected and rotated model  
236 and the data points. To our knowledge, the Hausdorff distance,  $d$ , is an appro-  
237 priate way of measuring the resemblance between a model shape and a shape on  
238 a 2D image (Huttenlocher et al., 1993). It is a measure of the distance between  
239 two sets, which can be defined as follows:

$$d(X, Y) = \max\left\{\sup_{x \in X} \inf_{y \in Y} \|x - y\|_2, \sup_{y \in Y} \inf_{x \in X} \|x - y\|_2\right\} \quad (3)$$

240 where  $\|\cdot\|_2$  is the Euclidean norm in  $\mathbb{R}^2$ ,  $X$  and  $Y$  are two (possibly infinite)  
 241 sets in  $\mathbb{R}^2$  and  $\sup$  and  $\inf$  are the supremum and infimum, respectively. In  
 242 our case,  $X$  is one of the projections of our polytope,  $P_k$ , and  $Y$  is the convex  
 243 hull  $H_k$  of the measured contour  $D_k$  for  $k \in \{1; 2\}$ . Hours et al. (2014) have  
 244 shown that the Hausdorff distance in such cases can be written in a simplified  
 245 form involving only the vertices  $V_k$  of the projection  $P_k$ , so that the Hausdorff  
 246 distance becomes (for  $k \in \{1; 2\}$ ):

$$d_k(V_k, H_k) = \max\left\{\max_{x \in V_k} \min_{y \in H_k} \|x - y\|_2, \max_{y \in H_k} \min_{x \in V_k} \|x - y\|_2\right\} \quad (4)$$

Note that in Eq. (4)  $\min$  and  $\max$  have replaced  $\inf$  and  $\sup$  in Eq. (3) since these two sets compared in this equation are non-empty and finite. However, Eq. (4) is not a smooth function and therefore leads to optimization problems that are cumbersome to solve (Lakhtin and Ushakov, 2005). Therefore, we propose to take an averaged version of the Hausdorff distance, which, accounting for both projections now, can be written as:

$$\tilde{d} = \sum_{k=1}^2 \left\{ \frac{1}{M_k} \sum_{i=1}^{M_k} \min_{y \in H_k} \|x_i - y\|_2 + \frac{1}{N_k} \sum_{j=1}^{N_k} \min_{x \in V_k} \|x - y_j\|_2 \right\} \quad (5)$$

247 where  $M_k$  is the number of vertices of  $P_k$  and  $N_k$  is the number of points in  $H_k$ .

#### 248 4.2. Definition and solution of the optimization problem

249 The estimates of the size vector  $\mathbf{t}$  and of the rotation  $\mathbf{R}$  are taken to be the  
 250 solutions of the following nonlinear and nonconvex optimization problem:

$$\min_{\mathbf{t}, \mathbf{R}} \tilde{d} \quad (6)$$

251 The rotation matrix is parameterized using quaternions,  $\mathbf{q}$ , which gives (Schmidt  
 252 and Niemann, 2001; Hours et al., 2014):

$$\mathbf{R}(\mathbf{q}) = \frac{1}{\|\mathbf{q}\|_2^2} \begin{pmatrix} r_1(\mathbf{q}) & r_2(\mathbf{q}) & r_3(\mathbf{q}) \end{pmatrix} \quad (7)$$

253 where

$$\begin{aligned} r_1(\mathbf{q}) &= \begin{pmatrix} q_0^2 + q_1^2 - q_2^2 - q_3^2 \\ 2(q_1q_2 + q_0q_3) \\ 2(q_1q_3 - q_0q_2) \end{pmatrix}, \quad r_2(\mathbf{q}) = \begin{pmatrix} 2(q_1q_2 - q_0q_3) \\ q_0^2 - q_1^2 + q_2^2 - q_3^2 \\ 2(q_2q_3 + q_0q_1) \end{pmatrix}, \\ r_3(\mathbf{q}) &= \begin{pmatrix} 2(q_1q_3 - q_0q_2) \\ 2(q_2q_3 - q_0q_1) \\ q_0^2 - q_1^2 - q_2^2 + q_3^2 \end{pmatrix} \end{aligned} \quad (8)$$

254 It can be shown that  $\mathbf{q} \neq 0$  and that each element lies within  $[-1, 1]$  (Hours et al.,  
 255 2014). Due to the nonlinearity and nonconvexity of the problem, we choose to  
 256 solve the optimization problem Eq. (6) for every region in the morphology map  
 257 separately by putting constraints on the scaling vector  $\mathbf{t}$ . These constraints can  
 258 be precomputed either numerically (as shown in Hours et al. (2014)) or ana-  
 259 lytically (as shown in Borchert and Sundmacher (2013); Singh and Ramkrishna  
 260 (2013)). Furthermore, we warm-start a nonlinear solver (IPOPT, see Wächter  
 261 (2006)) from different initial conditions for the rotation matrix  $\mathbf{R}$ . The number  
 262 of initial points that need to be considered for the rotation matrix in order to  
 263 arrive at a good reconstruction of the particle morphology is dependent on the  
 264 complexity of the crystal, its rotation, as well as on the optical properties of the  
 265 imaging system (the quality of the images). For the cases presented in Section 5  
 266 we found that using a linearly spaced grid of  $7 \times 7 \times 7$  points (in axis-angle  
 267 representation (Hornegger and Tomasi, 1999)) is sufficient to get a good recon-  
 268 struction. We arrived at this conclusion by performing a sensitivity analysis on

269 an exemplary crystal of  $\alpha$  L-glutamic acid. Every grid point is given as initial  
270 condition to IPOPT, which is run until a specified level of accuracy on a local  
271 optimum is met. In the end, the optimal fit is taken to be the best parameter  
272 estimate  $(\hat{\mathbf{t}}, \hat{\mathbf{R}})$  resulting in the lowest value of  $\tilde{d}$ . Figure 5 gives a schematic  
273 overview of the procedure. Thus, for  $n$  grid points in the discretization of the  
274 rotation matrix and  $m$  morphology regions,  $n \times m$  optimization problems need  
275 to be solved for every pair of measured projections, i.e., for every particle. To  
276 get a reconstruction of one typical crystal takes approximately one minute us-  
277 ing the presented methodology. The size of the crystal is finally obtained by  
278 multiplying  $\hat{\mathbf{t}}$  with the scaling constant  $s$ .

## 279 5. Results and discussion

280 In order to evaluate the particle sizing procedure a selection of five differ-  
281 ent organic substances (see Table 1) has been investigated both by generating  
282 artificial images and by taking actual photographs of crystals produced in ex-  
283 periments. The substances are acetaminophen, ascorbic acid, Ibuprofen and the  
284  $\alpha$  and  $\beta$  polymorphs of L-glutamic acid. The selection covers a wide range of  
285 different crystal shapes from needle like crystals with a very high aspect ratio to  
286 platelets and rather compact crystals with a low aspect ratio. For reconstructing  
287 simulated particles, a cuboidal polytope was considered as well.

### 288 5.1. Error metric

289 Two different error metrics have been used to evaluate the performance and  
290 accuracy of the optimization algorithm. For generated images, the input data  
291 in terms of the true size of all facets as the scaling vector  $\mathbf{t}$  was available, hence  
292 the ratio  $\hat{t}_i/t_i$  for each facet  $i$  can be calculated. In addition to the recalculation  
293 error one can define the reprojection error  $\epsilon_{\Pi}$ :

$$\epsilon_{\Pi} = \frac{\|\mathbf{H}_1 \setminus P_1\|_2 + \|\mathbf{H}_2 \setminus P_2\|_2}{\|P_1\|_2 + \|P_2\|_2} \quad (9)$$

294 This error measures the difference between the input projection and the calcu-  
295 lated reprojection. It can be considered a measure of the quality of fit.

### 296 5.2. Analysis of generated images

297 As a first step, artificially generated images of acetaminophen crystals be-  
298 longing to each of the five morphology regions (cf. Figure 2) have been studied.  
299 Secondly, artificial images of the other four crystal systems and of the cuboid  
300 model have been analyzed as well. The model parameters in terms of matrices  $\mathbf{A}$   
301 and  $\mathbf{M}$  are reported in Tables 2 to 7. The resulting images have been processed  
302 according to Section 3.3 and passed over to the optimization algorithm.

303 *5.2.1. Fitting of different morphologies of the same crystal*

304 The first series of simulations is aimed at proving that the optimization  
305 algorithm is capable of finding satisfactory results independent of the morphol-  
306 ogy region where the particle to be analyzed is located. In a case study for  
307 acetaminophen, for which the morphology map is reported in Figure 2, 100 ran-  
308 domly oriented polytopes for each of the five regions have been generated and  
309 handed over to the image analysis and size calculation procedures. In Figure 6,  
310 example polytopes for each region (black solid lines in the first column), their  
311 projections (dark grey areas), and input scaling vectors  $\mathbf{t}$  are reported. For  
312 each of the randomly selected original polytope the reconstructed polytope is  
313 reported as an overlay in red. As one can see, there is an excellent agreement  
314 between input and optimization result. The corresponding fitting contour is  
315 plotted on top of the input projection as red dashed lines and the estimated  
316 scaling vector  $\hat{\mathbf{t}}$  is listed next to the input. It is noteworthy that for regions a)  
317 and b) only two characteristic facet families can be identified because the third  
318 facet family is absent from the crystal morphology (which is correctly identified  
319 by the optimization algorithm). For cases where a facet is missing (i.e., regions  
320 a) and b)) the optimization algorithm returns the value at the boundary of  
321 the corresponding region. In the reported exemplary cases, the matching be-  
322 tween input and fitted polytope is excellent, and both scale and rotation have  
323 been calculated correctly. However, for some rotations of the same polytopes  
324 the estimated scaling vector deviates slightly from the input scaling vector. To  
325 quantify this effect, we report statistics of the scaling and reconstruction error  
326 for all 100 reconstructed particles in the last two columns of this figure. For  
327 the scaling statistics, we report the mean of the relative scale  $\hat{t}_i/t_i$  of the  $i$ -th  
328 characteristic facet family as a square. The error bars indicate the standard  
329 deviation. The reprojection error in the last column is plotted similarly. Ad-

330 ditionally, the morphology distribution is also reported as contour plots in a  
331 morphology map, cf. Figure 7. The scaling error distribution shows an overall  
332 low deviation of the estimated scale as compared to the input. In cases b) and  
333 d) the optimization result is almost perfect and contour lines mostly overlap  
334 with the input. Also in cases a) and c), the deviation does also not exceed 10%  
335 of the input and the mean of the scaling factors is correctly identified. For the  
336 platelet-like morphology of region e) the standard deviation for both the scal-  
337 ing and the reprojection error is larger, while the mean value is still correctly  
338 identified. This result reflects the fact that some morphology regions result in  
339 projections that are easy to analyze due to a high number of very characteristic  
340 facets, while in other cases the optimization is much more difficult as, depend-  
341 ing on the rotation, ambiguous solutions can be found. A clear indication for  
342 this behavior is the fact that even for a reprojection error of zero, the scaling  
343 error can be considerably larger. However in general, the procedure delivers  
344 promising results for the majority of cases studied and even in the worst case  
345 the results are still satisfactory, so that we are confident that one may obtain  
346 meaningful scaling information.

### 347 *5.2.2. Fitting of different crystal systems*

348 The size calculation procedure can easily be extended to different crystal  
349 shapes. Four additional organic crystal systems and a cube are selected as  
350 validation cases. The analysis protocol is similar to Section 5.2.1. 100 randomly  
351 oriented polytopes of known size have been generated, analyzed, and the results  
352 are presented in Figure 8. In this figure we report the same information as in  
353 Figure 6. Morphology maps, input scaling vectors (as crosses) and distributions  
354 of estimated polytopes (as red contour curves) are presented in Figures 9 to 11.  
355 In general, fitted crystals match the input projections very well, yielding only  
356 small reprojection errors. It is worth noting that the error in the scaling vector

357 for all considered facets is less than 3% in all cases but for  $\beta$  L-glutamic acid.  
358 However, this might be a result of the very small facets at the tip of the elongated  
359 crystal, which are obviously hard to detect in projections, even by the human  
360 eye. The reprojection errors are smaller than 3% in all cases except for the  $\alpha$  and  
361  $\beta$  polymorph of L-glutamic acid. Even in these worst case scenarios the error  
362 is below 8%. Considering the statistics for all 100 projected particles (last two  
363 columns), one can see that for all considered particle shapes, with the exception  
364 of ascorbic acid crystals, the crystal shape is on average correctly identified and  
365 the standard deviation in the identification is negligible, i.e., the results are  
366 again almost perfect. For the platelet-shaped ascorbic acid crystal the particle  
367 shape is on average also correctly identified, but the standard deviation (just  
368 as in the case of the acetaminophen morphology e)) is larger than in the other  
369 cases. This is an indication of the fact that the projections of a platelet-shaped  
370 object are strongly dependent on its alignment in space. This can result in cases  
371 where the vertices that are visible in a projection are not sufficient to uniquely  
372 determine the objects rotation, thus yielding non-perfect fits. These results are  
373 encouraging as the algorithm seems to exhibit a good performance for most  
374 crystal systems using simulated measurements.

### 375 *5.3. Importance of the stereo angle*

376 In order to assess the potential of the proposed method for observing sin-  
377 gular crystals in a stereoscopic hot stage setup the importance of the observation  
378 angle, i.e., the angle between the two cameras as illustrated in Figure 12, is  
379 investigated. Commercially available stereo microscopes have typically obser-  
380 vation angles between  $12^\circ$  and  $14^\circ$ . However, customized devices could feature  
381 stereo angles up to  $90^\circ$ . The change in the reconstruction performance is inves-  
382 tigated on the example of an Acetaminophen crystal with  $t_i = 1$  for all  $i$ , i.e., the  
383 same case as in Figure 6c). Stereo angles of  $12^\circ$ ,  $14^\circ$ ,  $30^\circ$ ,  $45^\circ$ ,  $60^\circ$ ,  $75^\circ$  and  $90^\circ$

384 have been investigated using 100 randomly orientated particles for each stereo  
385 angle. Note that the assumption of random orientation for a compact particle  
386 is a valid one, however, one can expect to obtain the same trend (see discussion  
387 below) for cases where the particle exhibits a preferential orientation due to its  
388 morphology. We are reporting the mean reprojection error and its standard de-  
389 viation in Figure 13(a). It is apparent from the very low mean reprojection error  
390 (square symbols in Figure 13(a)) that independent of the observation angle, the  
391 optimization algorithm is able to find a set of  $\hat{R}$  and  $\hat{t}$ , which is able to match  
392 the measured projections. Note that the underlying histograms of reprojection  
393 errors for each stereo angle are skewed towards low reprojection errors, so that  
394 the lower ends of the error bars (solids lines in Figure 13(a)) extend below zero.  
395 However, when considering the estimated crystal sizes,  $\hat{t}_i$ , one can see that nei-  
396 ther the accuracy nor the precision of the estimates are satisfactory at low stereo  
397 angles, which can be seen in Figure 13(b) where the exemplary statistics for  $\hat{t}_2$   
398 are reported. Only for stereo angles greater than  $60^\circ$  do the precision and the  
399 accuracy become satisfactory. This indicates that for small stereo angles, the  
400 restoration problem becomes less unique, i.e., the algorithm is likely to find a  $\hat{t}$ ,  
401 which is able to match the measured projection data well, but is far away from  
402 the actual crystal size  $t$ . Consequently, the reconstruction of a general faceted  
403 crystal from a single projection is a hopeless endeavor. Thus, the take-home  
404 message is that a custom made stereoscopic hot stage setup should therefore be  
405 built with a stereo angle of at least  $60^\circ$ .

#### 406 *5.4. Analysis of actual photos of crystals*

407 The second series of experiments is based on actual photographs of particles  
408 taken in the flow through cell, i.e., for orthogonal projections. Crystals of all  
409 five previously mentioned organic substances have been prepared according to  
410 the following procedure.

411 *5.4.1. Preparation of crystals*

412 Acetaminophen (Sigma Aldrich, purity > 98%) crystals were obtained by  
413 cooling a saturated water solution from 35°C to 5°C in 6 hours, resulting in a  
414 suspension that was subsequently wet sieved and dried (Kempkes et al., 2010b).  
415 The sieve fraction above 200  $\mu\text{m}$  was used for later analysis.

416 L-Ascorbic acid (Sigma Aldrich, reagent grade) was crystallized from a sat-  
417 urated methanol (Merck Chemicals, purity > 99.9%) solution by cooling the  
418 mixture and pestled seed from 35°C to 5° in 6 hours. The product was wet  
419 sieved and analyzed directly afterwards.

420 Ibuprofen (4-Isobutyl- $\alpha$ -methylphenylacetic acid, TCI Europe, purity > 98%)  
421 crystals were prepared by a temperature cycle approach from a saturated mix-  
422 ture of 50 wt% ethanol (Scharlab, HPLC grade) and 50 wt% water to which 8  
423 wt% (solvent basis) Pluronic F127 (Sigma Aldrich, purity > 99%) were added.  
424 The solution was cooled from 28°C to 1°C in 3.25 hours, heated up to 25°C in  
425 2.5 hours and cooled down again to 1°C in 4 hours. The product suspension  
426 was filtered and dried before analyzing.

427 Crystals of the  $\alpha$  polymorph of L-glutamic acid have been precipitated by  
428 mixing equimolar amounts of L-glutamic acid monosodium salt hydrate (Sigma  
429 Aldrich, purity > 99%) and hydrochloric acid (Fluka, 37%) in water and contin-  
430 uous stirring for one hour at 5°C (Cornel et al., 2009). The product suspension  
431 was split in two fractions. The first fraction was wet sieved to obtain crystals  
432 larger than 125  $\mu\text{m}$  which were directly analyzed afterwards. The other half  
433 was filtered and dried to be used in a transformation experiment to obtain  $\beta$   
434 L-Glutamic acid crystals. To this end, a saturated (with respect to the  $\alpha$  poly-  
435 morph) solution of L-glutamic acid monosodium salt hydrate (Sigma Aldrich,  
436 purity > 99%) and hydrochloric acid (Fluka, 37%) in water at 45°C was pre-  
437 pared.  $\alpha$  crystals were added and the solution was stirred for 48 hours to ensure

438 complete transformation. The product suspension was then directly analyzed.

439 Water used during experiments was always taken from a MilliQ Advantage  
440 A10 system (Merck Chemicals).

#### 441 *5.4.2. Analysis results for actual photographs*

442 After particle production was completed, crystals were suspended and pho-  
443 tographed in the flow through cell. Images have been processed to extract the  
444 sets  $D_1$  and  $D_2$  and furthermore  $H_1$  and  $H_2$ . Figure 14 shows exemplary im-  
445 ages, fitted contours, reconstructed polytopes, the resulting particle size, the  
446 re-projection error for the example case and the reprojction error distribution.  
447 As explained before, the calculated scaling vector  $\hat{\mathbf{t}}$  is multiplied with the scaling  
448 constant  $s$  thus yielding the size of the reconstructed polytope in  $\mu\text{m}$ . As the  
449 true vectors  $\mathbf{t}$  are unknown for these crystals, only the reprojction error  $\epsilon_{\Pi}$  can  
450 be reported. In general, the fitting error is in the same low range as for the  
451 simulated images demonstrated in the previous section. The reprojction error  
452 is again highest for ascorbic acid, just as in the case of the simulated particles.  
453 Additional to the reasons for non-perfect fits given for the simulated particles,  
454 one has to realize that real crystals are often not perfectly faceted objects. As  
455 such, the particle model may inherently not be able to represent crystals that  
456 have been photographed in the flow through cell. Additional noise is introduced  
457 due to dust on the images, errors in the contour extraction, a limited resolution,  
458 and a limited depth of field. Despite these limitations, the fitting results appear  
459 to be generally satisfying enough to extract useful facet information form the  
460 reported images.

#### 461 *5.4.3. Crystal Growth*

462 Finally, we want to investigate the possibility of applying the image analysis  
463 and polytope recognition procedure to a dynamic experiment in which crystals

464 grow. A population of Acetaminophen crystals is exposed to supersaturation  
465 during cooling crystallization in water, and images of particles in suspension are  
466 continuously recorded for the entire duration of the experiment.

467 Acetaminophen seeds were produced by rapidly cooling a saturated solution  
468 of acetaminophen and water (Sigma Aldrich, purity 98.0%) from 40°C to 10°C.  
469 The product was filtered, dried, and sieved. The sieve fraction between 125  
470 and 250  $\mu\text{m}$  was used. Crystals have been suspended in a saturated solution  
471 of acetaminophen and water at 30°C at a suspension density of 1 g/kg. The  
472 resulting suspension was then cooled to 25°C with a cooling rate of 0.5°C/h.  
473 The temperature was subsequently kept constant for additional 200 minutes.

474 The low cooling rate (and therefore low supersaturation level) causes the  
475 aspect ratio of the acetaminophen crystal population to increase, which is in  
476 agreement with previous observations (Kempkes et al., 2010a). With reference  
477 to Figure 2 the vast majority of particles was measured in region (c). The  
478 change of the distribution from the seed crystals (red) to the product crystals  
479 (blue) is shown as isosurface plots and isocontour plots in Figure 15. Note that  
480 information during the cooling crystallization experiment was recorded but not  
481 analyzed due to a lack of computational power, because the reconstruction of one  
482 crystal of typical complexity (i.e., obtaining a solution of the nonlinear problem  
483 Eq. (6) from all warm start points) takes roughly one minute. However, in  
484 order to obtain a smooth PSSD the analysis of several thousand particles is  
485 necessary, thus resulting in several days of computational time. The identical  
486 experiment has been presented before in Schorsch et al. (2014) using a different  
487 type of particle model. In contrast to the generic particle analysis the polytope  
488 based approach delivers very scattered distributions. It is noticeable that the  
489 movement of the center of the PSD is only due to an increase in the scaling  
490 factors  $s$  and cannot be attributed to a dominant change in  $t$ , i.e., the before-

491 mentioned increase in aspect ratio could not be clearly identified. Since the  
492 computational cost of the polytope reconstruction in comparison to the generic  
493 particle approach is high, a thorough comparison between the two methods  
494 based idealized pictures seems to be warranted and is presented in the following  
495 section.

### 496 *5.5. Comparison to Generic Particles*

497 In previous works (Schorsch et al., 2012, 2014) we have reported a set of  
498 generic particle models to describe crystals. These models are spheres, cylin-  
499 ders and cuboids; as such these generic particles are able to cover a wide range  
500 of prototypical shapes. The size information in this case is computed from geo-  
501 metric information that is available in a stereoscopic setup, but without directly  
502 minimizing the reprojection error. Clearly, the additional computational effort  
503 of the polytope reconstruction procedure needs to be justified. A direct compar-  
504 ison between the two techniques is difficult because the output of the two models  
505 is quite different (generic particle vs. faceted polytope). Hence, we compare the  
506 two most important macroscopic properties of the crystals, i.e., their aspect  
507 ratio (i.e., the ratio of maximum to minimum size) and the particle volume.  
508 The analysis is performed using an exemplary particle of each of the previously  
509 mentioned organic substances and a cube-shaped particle, which were randomly  
510 oriented 100 times (each). The input size vectors, estimated volumes and aspect  
511 ratios and the true aspect ratio are reported in Table 8 for each of these cases.  
512 The outcome of this comparison depends on the chosen inputs, i.e., one may  
513 find conditions for which either approach is better. Therefore, we have chosen  
514 cases which cover a reasonable range of different geometries.

515 For the needle-like  $\beta$  glutamic acid and the cubes, there is no advantage in  
516 using the polytope model because the shape gets almost perfectly matched in  
517 all cases independent of the particle model. For the other cases in average the

518 generic case delivers a reasonable estimation. However the standard deviation  
519 of the underlying recovered distribution is wider than for the polytope recon-  
520 struction, which means that generic models sometimes over- or underestimates  
521 the particle size due to an inherent mismatch between generated particle and  
522 fitted shape. The platelet-like ascorbic acid crystals are not captured well by  
523 either particle model, which is not surprising for the reasons mentioned already  
524 above. It is worth noting that the set of generic particle shapes can also be  
525 easily extended to get a better model representation for a specific crystal shape  
526 if required, so as to improve the reconstruction accuracy, while preventing an  
527 increased computational effort.

528       Calculating the faceted shape of a crystal requires particles of a certain size  
529 or an imaging system with an excellent resolution that equals the quality of a  
530 microscope in order to have sufficient contour data to solve the polytope recon-  
531 struction problem. It also requires crystals of a very high quality, i.e., crystals  
532 without damaged vertices, etc., to obtain meaningful results. Given that many  
533 crystals obtained from a real crystallizer do exhibit some kind of imperfection,  
534 the advantage that a polytopic model describes the crystal morphology of pri-  
535 mary particles on a first principles level vanishes when moving from idealized  
536 generated images to real crystals produced in real crystallization experiments.  
537 The generic particle classes on the other hand are more robust and deliver good  
538 estimates of the most important macroscopic properties (aspect ratio and vol-  
539 ume of the crystal) at a much lower computational effort.

## 540 **6. Concluding Remarks**

541       An optimization-based approach is applied to reconstruct the faceted mor-  
542 phology of crystals from its two stereoscopic projections. These projections are  
543 obtained from two different but coplanar observation directions. To investigate

544 the influence of the geometry of the measurement device and the crystal's shape,  
545 we have varied the angle between the two coplanar observation directions (the  
546 so-called stereo angle) and the orientation and morphology of the analyzed par-  
547 ticles. Using simulated images, we have demonstrated that the approach works  
548 satisfactorily with a stereo angle of  $90^\circ$  by using simulated images of crystals  
549 of five organic substances. In these cases, the procedure delivers nearly perfect  
550 reconstructions of the input projections, except for rare cases where particles  
551 are seen from unfavorable angles where only an insufficient number of vertices  
552 is visible or where the facets themselves were rather small. A rotational de-  
553 pendence of the reconstruction quality especially noticeable for platelet shaped  
554 particles. However, even in these worst cases the error in the estimated sizes  
555 never exceeded 10%.

556 In our investigation of different stereo angles, we have found that a mini-  
557 mum of  $60^\circ$  is required in order to obtain trustworthy facet size information  
558 using our optimization based approach. It can be argued that the information  
559 content of the two projections decreases the smaller the stereo angle becomes,  
560 so that the uniqueness of the solution to the optimization problem disappears,  
561 i.e., a number of different orientations and size vectors yield indistinguishably  
562 similar projections when they are compared to the measured projections. Con-  
563 sequently, a stereoscopic hot-stage apparatus that can be used to obtain high  
564 quality information about the faceted morphology of a crystal should be con-  
565 structed with a stereo angle of at least  $60^\circ$ .

566 Comparing calculated reprojections with photographs of crystals from a real  
567 crystallization process (obtained using the setup presented in Schorsch et al.  
568 (2014)) yields a satisfactory agreement with respect to the reprojection error as  
569 well, however the correctness of the estimated facet sizes appears questionable,  
570 since a shape change of the crystals that occurred in a cooling crystallization ex-

571 periment of Acetaminophen crystals, that was clearly identifiable using generic  
572 model particles and by a visual comparison of initial and final particles, could  
573 not be identified anymore. We believe that the growing crystals from a real crys-  
574 tallization experiment carried out in a crystallizer are often of too low quality to  
575 accurately and convincingly reconstruct a faceted morphology from two projec-  
576 tions obtained of such particles. Therefore, we find that, at least for an on-line  
577 measurement technique such as the flow through cell, generic particle models  
578 yield a reasonable estimate of shape and size distributions for many applications  
579 at a much lower computational cost that enables the use of the measurement  
580 device for control purposes. Nevertheless, we think that a possible area of appli-  
581 cation for the proposed polytopic reconstruction method could be in hot stage  
582 microscopy where the crystals grown are typically of higher quality with fewer  
583 imperfections.

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587 **Notation**

$A$	Matrix of normal vectors $\in \mathbb{R}^{3 \times m}$	[-]
$\mathbf{a}_i$	Normal vector of the $i$ -facet $\in \mathbb{R}^3$	[-]
$C$	Crystal polytope	[-]
$D_{1/2}$	Dataset $\in \mathbb{R}^2$	[-]
$d_i$	2D data point	[-]
$M$	Growth dependency matrix $\in \mathbb{R}^{m \times p}$	[-]
$H_1, H_2$	Convex hull of a 2D dataset	[-]
$i$	Index	[-]
$M$	Number of vertices of a polytope projection	[-]
$m$	number of facets considered in a model	[-]
$N$	Number of points on the convex hull of a projection	[-]
$p$	number of independent growth directions	[-]
$R$	rotation matrix $\in SO_3(\mathbb{R})$	[-]
$s$	scaling constant	$[\mu\text{m}/\text{pixel}]$
$\mathbf{t}$	scaling vector $\in \mathbb{R}^p$	[-]
$V_i$	Matrix of vertices of a polytope $\in \mathbb{R}^{3 \times p}$	[-]
$D_i$	dataset $\in \mathbb{R}^2$	[-]
$x$	Cartesian coordinates $\in \mathbb{R}^2$ or $\mathbb{R}^3$	[-]
$Y$	data set $\in \mathbb{R}^2$	[-]
$d$	Hausdorff distance	[pixel]
$\tilde{d}$	Averaged Hausdorff distance	[pixel]
$\epsilon_{\Pi}$	reprojection error	[-]
$P_1, P_2$	2D projection of a polytope	[-]

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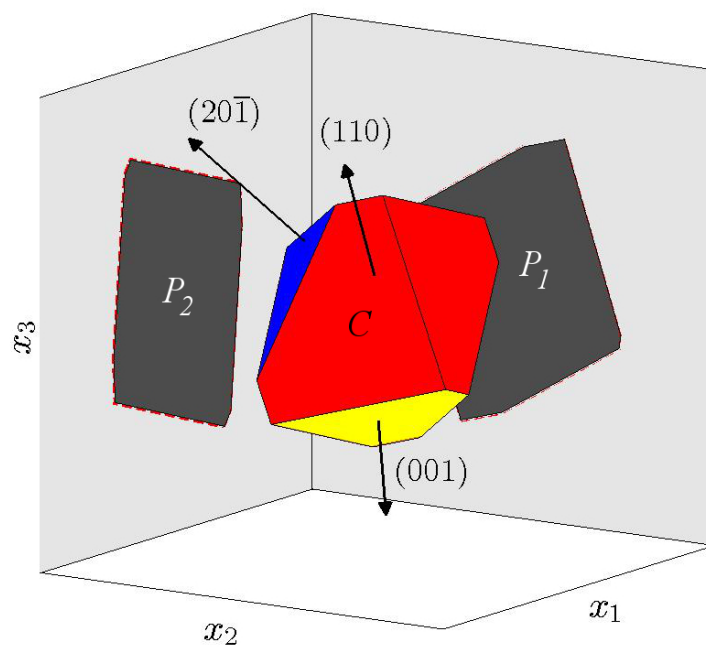


Figure 1: Projections, fitted contours, and 3D representation of a polytope of a simulated acetaminophen crystal. Projections: dark grey areas, boundary contours: dashed red lines, reconstructed polytope: red, blue, and yellow body.

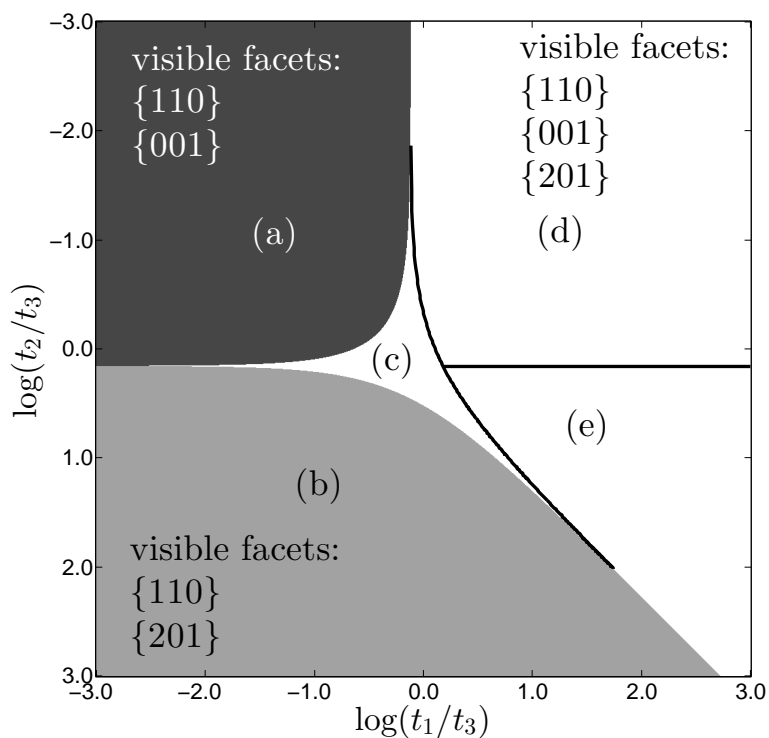


Figure 2: Morphology map of acetaminophen with facet groups reported detailed in Table 2. Different grey values indicate morphology domains with a common set of constituting facets. Within the white area, two black lines separate regions of different sets of characteristic edges. Example polytopes, indicated as a)-e), are shown in Figure 3.

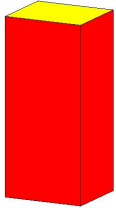

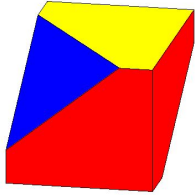
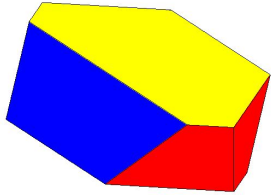
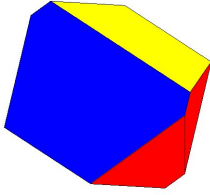
region	sample Polytope	t
region (a)		0.21 0.55 1.00
region (b)		0.45 4.48 1.00
region (c)		1.00 1.00 1.00
region (d)		2.31 0.67 1.00
region (e)		4.36 2.18 1.00

Figure 3: Sample polytopes of all five morphology regions of Acetaminophen corresponding to the regions in Figure 2 and the scaling vectors used to construct these. The  $\{110\}$  family is drawn in red,  $\{001\}$  in yellow and  $\{20\bar{1}\}$  in blue.

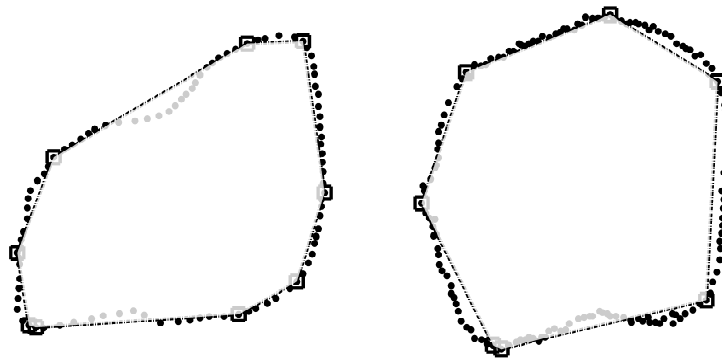


Figure 4: Output of the Douglas-Peucker applied to a pair of images of an acetaminophen crystal. The points extracted from the images appear as black dots. The black squares correspond to the vertices of the polygonal line produced by the Douglas-Peucker algorithm. The convex polygon corresponds to the convex hull of the output points of the Douglas-Peucker procedure.

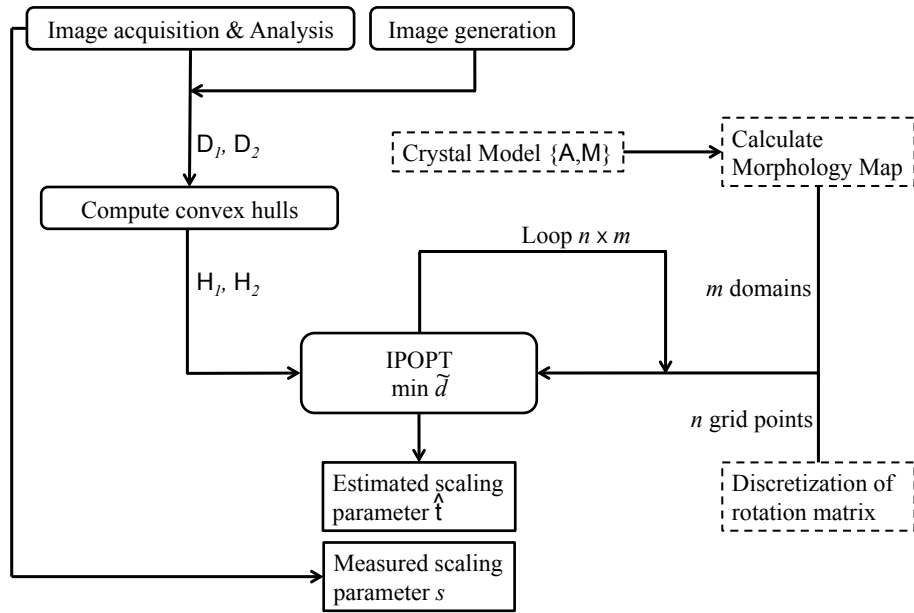


Figure 5: Schematic overview of the optimization algorithm.

	random sample particle			statistics for 100 particles			
	3D polytopes	Projections	$\mathbf{t}$	$\hat{\mathbf{t}}$	$\epsilon_{\Pi}$	scaling error (normalized)	reprojection error
region (a)			0.21 0.55 1.00	0.21 0.55 N/A	0.00		
region (b)			0.45 4.48 1.00	0.45 N/A 1.01	0.01		
region (c)			1.00 1.00 1.00	1.00 1.00 1.00	0.00		
region (d)			2.31 0.67 1.00	2.30 0.66 1.00	0.01		
region (e)			4.36 2.18 1.00	4.36 2.18 1.00	0.00		

Figure 6: Reconstruction of faceted polytopes for all five morphology regions of acetaminophen. From left to right: Three dimensional polytopes of the input data (solid black lines) and the reconstructed polytope (dashed red lines), orthogonal two dimensional projections of the input polytope (dark gray area) and the fitted polytope (red dashed line), input scaling vector  $\mathbf{t}$ , calculated scaling vector  $\hat{\mathbf{t}}$ , error between input and calculated scaling vectors, reprojection error, error distribution of the scaling vector, error distribution of the reprojection error. Note that for regions a and b, only two entries of the scaling vector  $\mathbf{t}$  can be estimated as the third facet family was detected to be not visible.

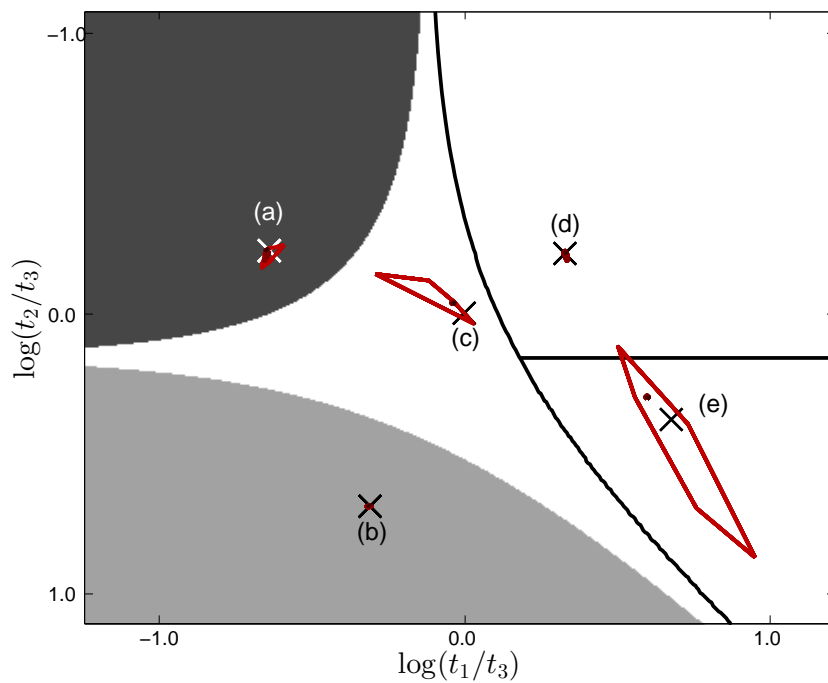


Figure 7: Zoom in of the morphology map for acetaminophen and the position of the 5 studied cases of the different morphology regions as black/white crosses. The distributions of the best fitting  $\hat{t}$  of the five case studies as indicated in Figure 6 are plotted as red contour curves into the morphology plane at levels of 1%, 2% and 50% of the maximum of the distribution. In cases in which only two indices of  $\hat{t}$  can be obtained (due to some facets being only virtual), the value of  $\hat{t}_i$  at the boundary of the morphology cone has been used to calculate the ratio.

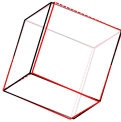

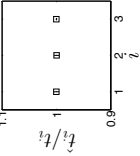
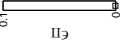
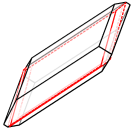

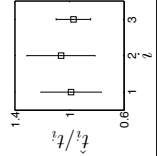
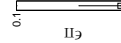
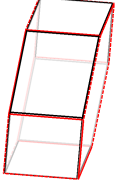

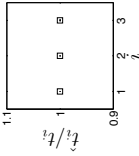
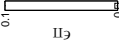
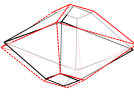

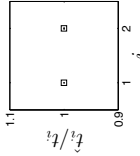
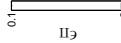
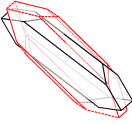
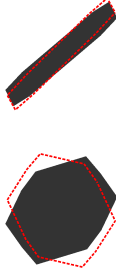
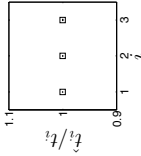
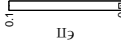
	randomly sampled particle			statistics for 100 particles			
	3D polytopes	Projections	$\mathbf{t}$	$\hat{\mathbf{t}}$	$\epsilon_{\Pi}$	scaling error (normalized)	reprojection error
Cube			1.00 1.00 1.00	1.01 1.03 0.97	0.01		
Ascorbic Acid			1.00 2.00 4.00	0.93 2.10 3.83	0.07		
Ibuprofen			2.00 2.00 3.00	2.07 1.95 3.01	0.03		
$\alpha$ -L-glutamic			1.00 1.30	1.06 1.29	0.00		
$\beta$ -L-glutamic			1.00 2.00 5.00	0.91 2.19 5.00	0.07		

Figure 8: Fitting of randomly selected examples of four different crystal systems and a cube. From left to right: Three dimensional polytopes of the input data (solid black lines) and the reconstructed polytope (dashed red lines), orthogonal two dimensional projections of the input polytope (dark gray area) and the fitted polytope (red dashed line), input scaling vector  $\mathbf{t}$ , calculated scaling vector  $\hat{\mathbf{t}}$ , error between input and calculated scaling vectors, reprojection error, error distribution of the scaling vector, error distribution of the reprojection error.

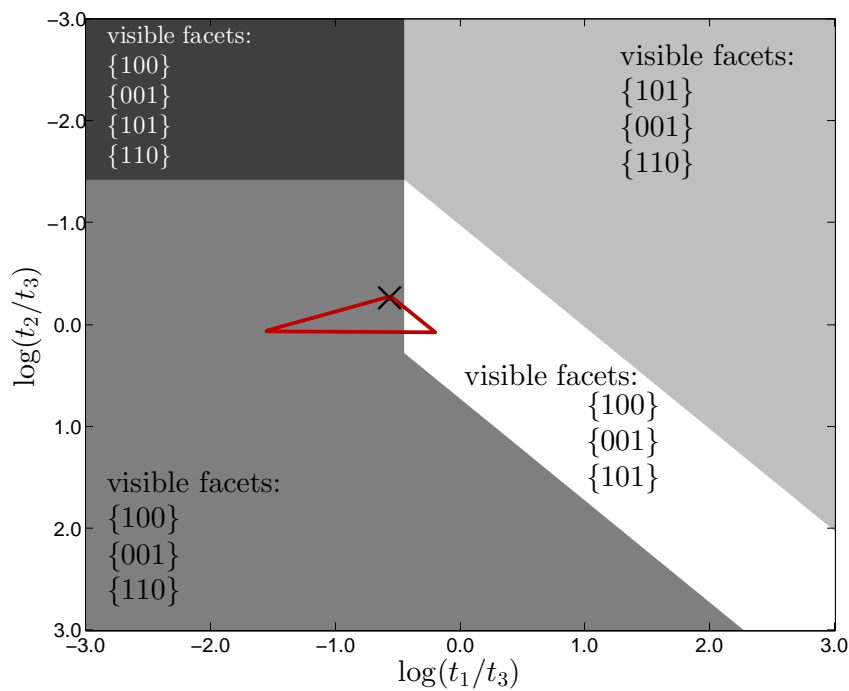


Figure 9: Morphology map for Ascorbic Acid and the position of the reconstruction case study of indicated as black cross. The distribution of the best fitting  $\hat{t}$  of the as listed in Figure 8 is plotted as red contour curves into the morphology plane at levels of 1%, 2% and 50% of the maximum of the distribution. In cases in which only two indices of  $\hat{t}$  can be obtained (due to some facets being only virtual), the value of  $\hat{t}_i$  at the boundary of the morphology cone has been used to calculate the ratio.

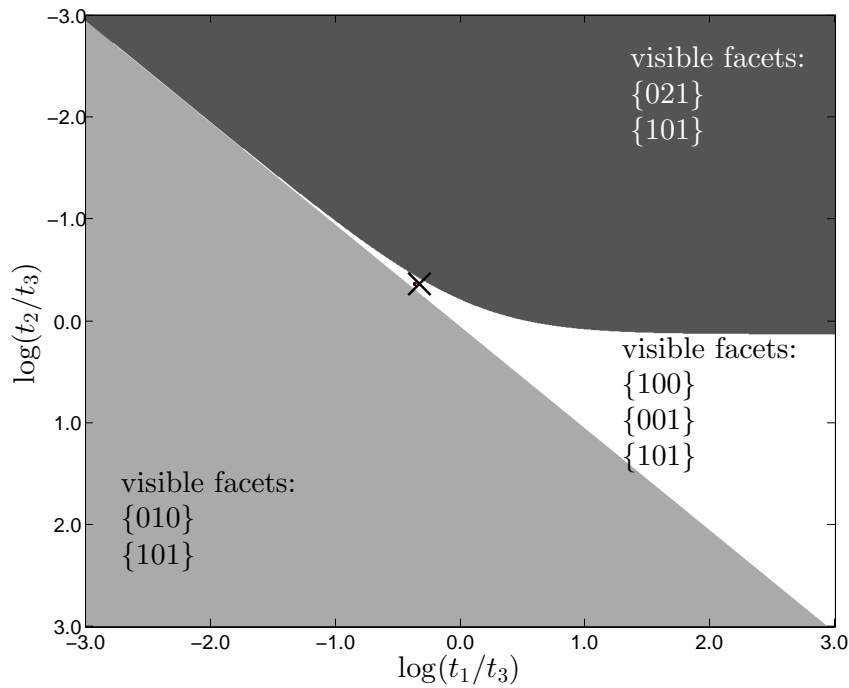


Figure 10: Morphology map for  $\beta$  L-Glutamic Acid and the position of the reconstruction case study of indicated as black cross. The distribution of the best fitting  $\hat{t}$  of the as listed in Figure 8 is plotted as red contour curves into the morphology plane at levels of 1%, 2% and 50% of the maximum of the distribution. In cases in which only two indices of  $\hat{t}$  can be obtained (due to some facets being only virtual), the value of  $\hat{t}_i$  at the boundary of the morphology cone has been used to calculate the ratio. Due to the small variance in the reconstruction results in this case, contour lines do mostly overlap.

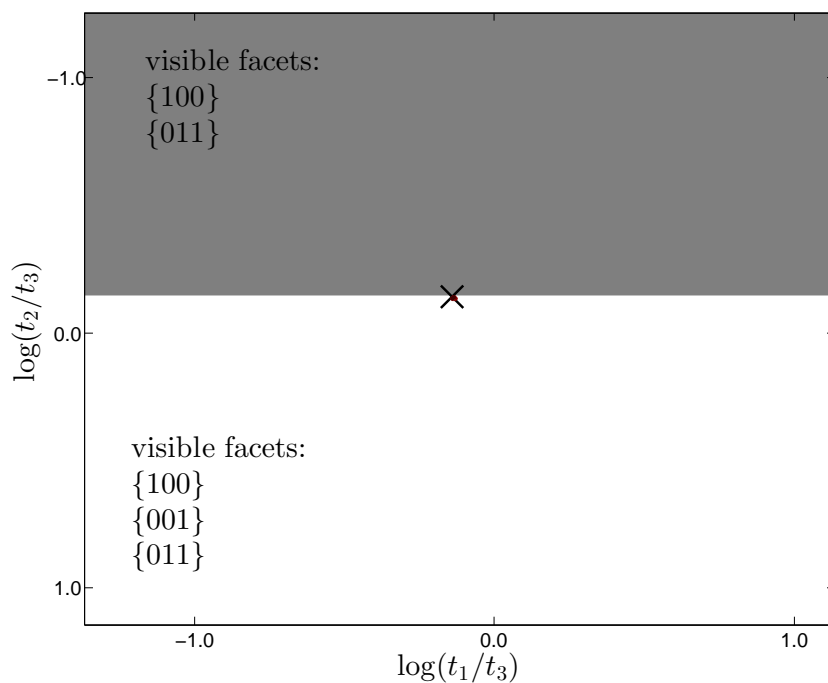


Figure 11: Morphology map for Ibuprofen and the position of the reconstruction case study of indicated as black cross. The distribution of the best fitting  $\hat{t}$  of the as listed in Figure 8 is plotted as red contour curves into the morphology plane at levels of 1%, 2% and 50% of the maximum of the distribution. In cases in which only two indices of  $\hat{t}$  can be obtained (due to some facets being only virtual), the value of  $\hat{t}_i$  at the boundary of the morphology cone has been used to calculate the ratio. Due to the small variance in the reconstruction results in this case, contour lines do mostly overlap.

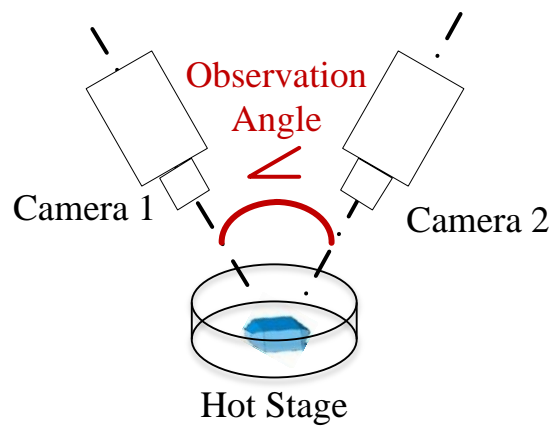
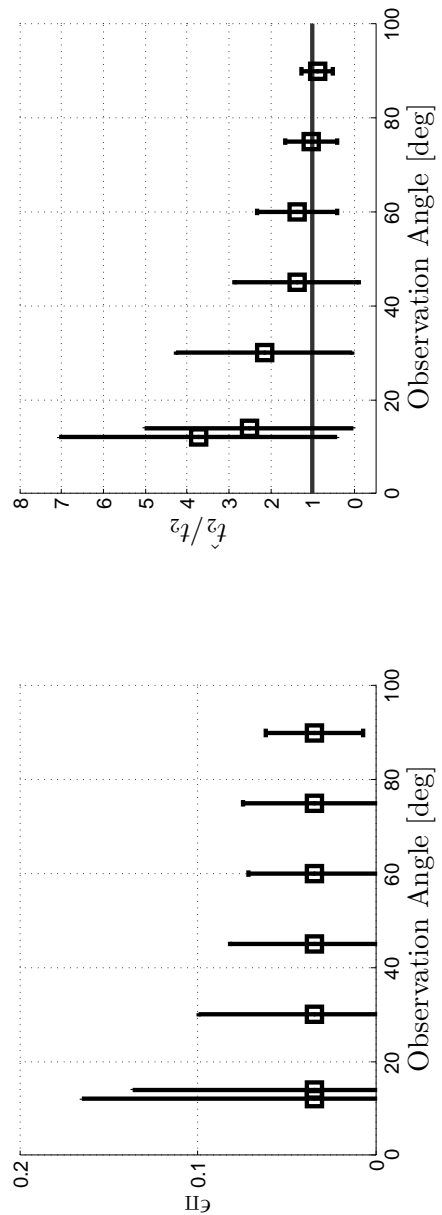


Figure 12: Explanation of the meaning of the observation angle for stereo microscopic hot stage studies.



(a)

(b)

Figure 13: Statistics for 100 randomly oriented Acetaminophen crystals of case (c) from Figure 6 when analyzed using different observation angles. (a) Dependency of the projection error on the observation angle in comparison to the reference case of an orthogonal camera setting (90°) and (b) accuracy of the estimated sizes to the input as a function of the observation angle. Squares indicate the position of the mean value and error bars represent the standard deviation of the distributions

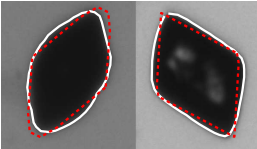
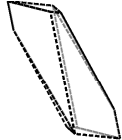
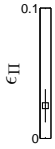
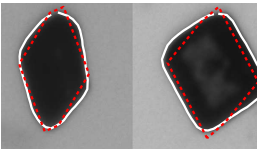
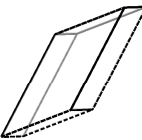

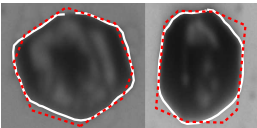
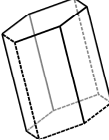
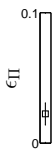
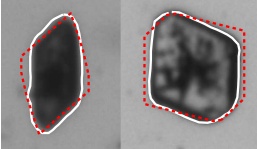
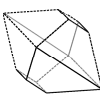

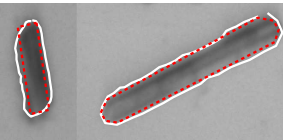

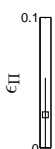
	random sample particle				statistics for 100 particles
	Projections	3D polytope	$s\hat{t}$	$\epsilon_{\Pi}$	$\epsilon_{\Pi}$ statistics
Acetaminophen			209 90 89	0.03	
Ascorbic			651 13 89	0.10	
Ibuprofen			266 193 210	0.03	
$\alpha$ L-glutamic			218 263	0.04	
$\beta$ L-glutamic			827 63 833	0.09	

Figure 14: Fitting of different organic crystals. From left to right: Photographs with extracted contours (white) and fitted projections (dashed, red), reconstructed 3D polytope, calculated scaling vector  $\hat{t}$ , reprojection error.

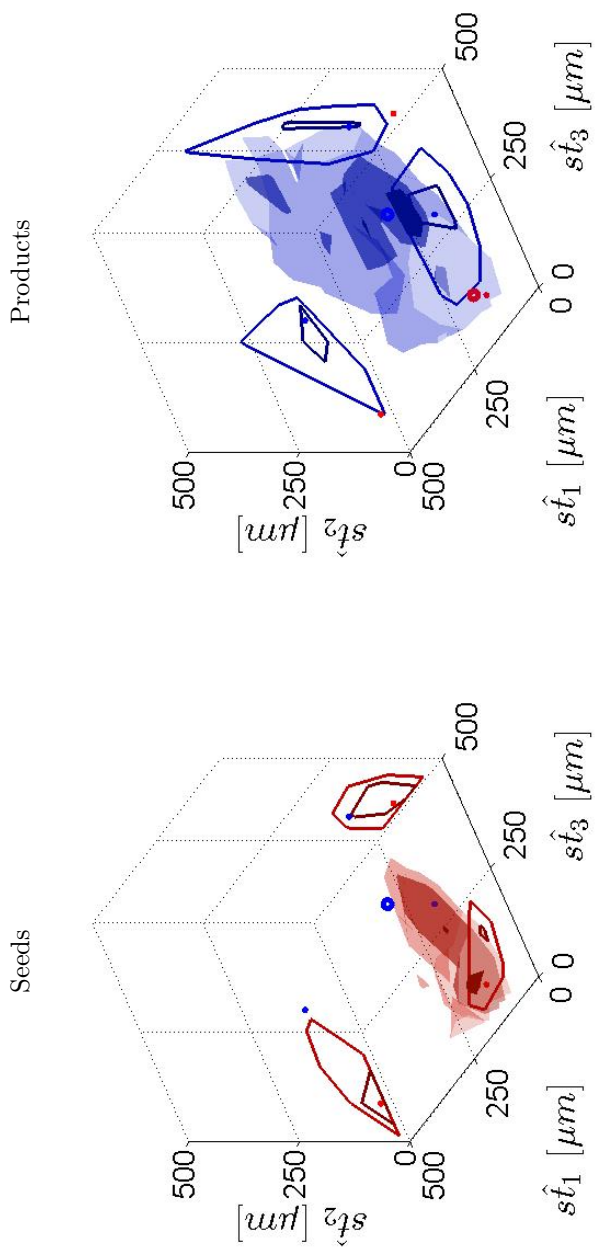


Figure 15: Contour surface plots of the number weighted 3D PSD of acetaminophen crystals measured during seeded cooling crystallization in water. Contour surfaces are drawn at values 0.15, 0.55, and 0.75 of the maximum of the PSD at the beginning (seeds, red) and end (products, blue) of the experiment. The average  $t_i$  of the distributions are plotted as red (seeds) and blue (products) circles into the  $(t_1 t_2)$ ,  $(t_1 t_3)$ ,  $(t_2 t_3)$  planes. Additionally the contour line plots of the distributions are plotted into each plane at values of 0.05 and 0.085.

Table 1: Overview of compounds and crystal forms considered in this paper

Substance	Space Group	Unit cell	a [Å]	b [Å]	c [Å]	$\beta$ [°]	References
Acetaminophen	P2 <sub>1</sub> /a	monoclinic	12.93	9.40	7.10	115.9	Haisa et al. (1976); Kuvadia and Doherty (2011)
Ascorbic Acid	P2 <sub>1</sub>	monoclinic	17.30	6.35	6.41	102.11	Hvoslef (1968); Srinivasan and Vanitha Devi (2010)
Ibuprofen	P2 <sub>1</sub> /c	monoclinic	14.67	7.89	10.73	99.36	Shankland et al. (1997); Cano et al. (2001)
$\alpha$ L-Glutamic Acid	P2 <sub>1</sub> 2 <sub>1</sub> 2 <sub>1</sub>	orthorhombic	7.06	10.3	8.75	90	Hirokawa (1955); Srinivasan and Dhanasekaran (2011)
$\beta$ L-Glutamic Acid	P2 <sub>1</sub> 2 <sub>1</sub> 2 <sub>1</sub>	orthorhombic	5.17	17.34	6.95	90	Hirokawa (1955); Srinivasan and Dhanasekaran (2011)

Table 2: Acetaminophen model data

Facet	$\mathbf{a}_i$			$\mathbf{M}$		
(110)	0.52	0.85	0	1	0	0
( $\bar{1}\bar{1}0$ )	0.52	-0.85	0	1	0	0
( $\bar{1}\bar{1}0$ )	-0.52	-0.85	0	1	0	0
( $\bar{1}10$ )	-0.52	0.85	0	1	0	0
(001)	0.26	0	0.97	0	1	0
(00 $\bar{1}$ )	-0.26	0	-0.97	0	1	0
(20 $\bar{1}$ )	0.69	0	-0.72	0	0	1
( $\bar{2}01$ )	-0.69	0	0.72	0	0	1

Table 3: Ascorbic Acid model data

Facet	$\mathbf{a}_i$			$\mathbf{M}$		
(100)	1	0	0	1	0	0
( $\bar{1}00$ )	-1	0	0	1	0	0
(001)	0.08	0	0.99	0	1	0
(00 $\bar{1}$ )	-0.08	0	-0.99	0	1	0
(10 $\bar{1}$ )	0.29	0	-0.96	0	1	0
( $\bar{1}01$ )	-0.29	0	0.96	0	1	0
( $\bar{1}\bar{1}0$ )	0.35	-0.94	0	0	0	1
( $\bar{1}10$ )	-0.35	0.94	0	0	0	1
(110)	0.35	0.94	0	0	0	1
( $\bar{1}\bar{1}0$ )	-0.35	-0.94	0	0	0	1

Table 4: Cube model data

Facet	$\mathbf{a}_i$			$\mathbf{M}$		
(100)	1	0	0	1	0	0
( $\bar{1}00$ )	-1	0	0	1	0	0
(010)	0	1	0	0	1	0
(0 $\bar{1}0$ )	0	-1	0	0	1	0
(001)	0	0	-1	0	0	1
(00 $\bar{1}$ )	0	0	-1	0	0	1

Table 5: Ibuprofen model data

Facet	$\mathbf{a}_i$			$\mathbf{M}$		
(100)	1	0	0	1	0	0
( $\bar{1}00$ )	-1	0	0	1	0	0
(001)	0.12	0	0.99	0	1	0
(00 $\bar{1}$ )	-0.12	0	-0.99	0	1	0
(011)	0.09	0.8	0.71	0	0	1
(0 $\bar{1}\bar{1}$ )	-0.09	0.8	-0.71	0	0	1
(0 $\bar{1}\bar{1}$ )	0.09	-0.8	0.71	0	0	1
(0 $\bar{1}\bar{1}$ )	-0.09	-0.8	-0.71	0	0	1

Table 6: L-Glutamic acid  $\alpha$  data

Facet	$\mathbf{a}_i$			$\mathbf{M}$	
(001)	0	0	1	1	0
(00 $\bar{1}$ )	0	0	-1	1	0
(110)	0.78	0.63	0	0	1
( $\bar{1}\bar{1}$ 0)	0.78	-0.63	0	0	1
( $\bar{1}$ 10)	-0.78	0.63	0	0	1
(1 $\bar{1}$ 0)	-0.78	-0.63	0	0	1

Table 7: L-Glutamic acid  $\beta$  data

Facet	$\mathbf{a}_i$			$\mathbf{M}$		
(010)	0	1	0	1	0	0
(0 $\bar{1}$ 0)	0	-1	0	1	0	0
(021)	0	0.89	0.45	0	1	0
(0 $\bar{2}$ 1)	0	-0.89	0.45	0	1	0
(02 $\bar{1}$ )	0	0.89	-0.45	0	1	0
(0 $\bar{2}$ $\bar{1}$ )	0	-0.89	-0.45	0	1	0
(101)	0.80	0	0.60	0	0	1
( $\bar{1}$ 01)	0.80	0	-0.60	0	0	1
(10 $\bar{1}$ )	0.80	0	-0.60	0	0	1
( $\bar{1}$ 0 $\bar{1}$ )	-0.80	0	-0.60	0	0	1

Table 8: Comparison of the fitting quality between Polytope and Generic Particle Models as ratio of the estimated and true particle volume and a comparison between the estimated and true aspect ratio (AR) as mean and standard deviation of 100 repetitions of fitting the respective crystal to the polytope particle and generic particle model each.

	input $t$		average est. volume/true volume		average est. AR		true AR
			Polytope	Generic	Polytope	Generic	
Cube	1	1	$1.00 \pm 3.8e-4$	$1.00 \pm 1.5e-15$	$1.00 \pm 0.01$	$1.00 \pm 1.2e-13$	1
Acetaminophen	2	1	$0.99 \pm 0.02$	$1.04 \pm 0.19$	$2.03 \pm 0.09$	$1.97 \pm 0.59$	2
Ascorbic Acid	1	2	$0.96 \pm 0.16$	$0.89 \pm 0.22$	$5.37 \pm 7.59$	$4.30 \pm 1.35$	4
Ibuprofen	2	2	$0.95 \pm 0.23$	$0.95 \pm 0.30$	$1.50 \pm 5.9e-5$	$1.65 \pm 0.58$	1.5
L-Glutamic Acid $\alpha$	1	1.3	$1.00 \pm 2.5e-4$	$0.96 \pm 0.29$	$1.30 \pm 1.5e-4$	$1.50 \pm 0.38$	1.3
L-Glutamic Acid $\beta$	1	10	$1.00 \pm 0.02$	$1.03 \pm 0.07$	$9.99 \pm 0.00$	$9.91 \pm 0.60$	10