

1 **Broadband Diffusing Wave Spectroscopy reveals**  
2 **microstructuring of polymer-drug system**

3 *Sandra Jankovic, Simone Aleandri, Mathias Reufer, Martin Kuentz\**

4

5 AUTHOR ADDRESS: *University of Applied Sciences and Arts Northwestern Switzerland,*  
6 *Institute of Pharma Technology, Muttenz, CH-4132 Switzerland*

7

8 KEYWORDS: macromolecular structuring, phase transition, broadband rheology, drug- polymer  
9 interaction

10

11 **ABSTRACT**

12 Microstructuring during a phase transition and crystallization in particular is critical for the  
13 physico-chemical properties of polymeric drug carriers and of the final dosage form. Extensive  
14 research has been dedicated to study polymeric matrices in drug delivery and despite of substantial  
15 progress, there are still unmet challenges such as a non-invasive mechanical analysis since  
16 classical rheological methods typically disturb the samples especially during a phase transition.  
17 This paper employs Diffusing Wave Spectroscopy (DWS) over a broad frequency band to study  
18 polymer-drug systems in a non-invasive way. Eutectic mixtures of polyethylene glycol (PEG) were  
19 investigated using two model drugs. While fenofibrate was barely interacting with the polymer,  
20 flurbiprofen provided a compound showing distinct molecular interactions with the carrier.  
21 Mechanical spectra were obtained during cooling of the molten polymer-drug systems. In  
22 conclusion, broadband DWS provided a better mechanistic understanding of the polymer-drug  
23 interactions and of macromolecular structuring during cooling of the eutectic melts. Such findings  
24 are relevant for a rationale design of pharmaceutical formulations during development and such  
25 knowledge would be also important for manufacturing to achieve drug products with reproducible  
26 quality characteristics.

27

28

29

30

31

## 32 INTRODUCTION

33 The rheological properties of pharmaceutically important materials such as polymer melts,  
34 colloids, gels, and dispersions are central to many fields of formulation development and  
35 manufacturing. Most mixtures exhibit complex rheological behavior since formulations include  
36 abundant polymers of different kinds.<sup>1</sup> Polymers that melt into a liquid are called to be  
37 thermoplastic and they are often processed by extrusion or molding. Polymeric crystallization may  
38 start when nuclei develop in a stochastic way and grow to a critical size in the cooling melt. Studies  
39 over the past two decades have provided important information on shear-induced crystallization  
40 and anisotropy in the direction of shear and show that crystallization proceeds as the nuclei grow  
41 into crystallites until all the melt has solidified.<sup>2</sup> When neighboring crystallites grow and segments  
42 of the chains forming these crystallites can no longer be incorporated into the crystalline domains,  
43 then amorphous regions start to form. This process is depending on the cooling rate so that a low  
44 cooling rate grants polymer chains more time to arrange or incorporate themselves into  
45 crystallites.<sup>2</sup>

46  
47 Crystallization in synthetic polymers typically produces polycrystalline aggregates that are called  
48 spherulites given their spherical morphology. These spherulites are radially symmetric arrays of  
49 fibrillary crystallites ranging in diameter from less than one micron to several millimeters.<sup>3</sup> In  
50 pharmaceutical systems, previous research revealed that interactions between the polymer and a  
51 drug can influence the crystallization behavior of the polymeric matrix.<sup>4</sup> Indeed, molecular drug-  
52 polymer interaction can occur already in the molten state and may not be easily evidenced in the  
53 solid state.<sup>5,6</sup> Especially difficult is to study the crystallization process because such a phase  
54 transition is easily perturbed by the analytical method thereby leading to experimental bias.

55 Mechanical rheology has a long tradition in the study of polymer melts.<sup>7</sup> Shear stress controlled  
56 rheometers with high sensitivity allow measurement in the linear visco-elastic regime (LVR).  
57 Current rheological research on polymer crystallization attempts to combine the mechanical  
58 experiment with monitoring techniques such as nuclear magnetic resonance (NMR), small-angle  
59 X-ray scattering or dynamic optical microscopy.<sup>8</sup> However, any mechanical rheometry is not  
60 contact-free and it can be problematic to measure within the LVR when very dynamic changes  
61 occur such as during polymer crystallization. Shear stress controlled oscillatory measurements are  
62 further limited because of a narrow range of accessible high frequencies (with a maximum of about  
63 100 Hz) and most important for the herein system, by the flow induced crystallization.<sup>9,10</sup>

64  
65 Modern microrheology based on Diffusing Wave Spectroscopy (DWS) has significantly extended  
66 the range of experimentally accessible frequencies and it is possible to measure non-ergodic  
67 samples such as gels and semi-crystalline materials in a non-invasive way.<sup>11-13</sup> DWS has been used  
68 previously in the field of Pharmaceutics to study self-emulsifying formulations, emulsions as well  
69 as solid drug dispersions.<sup>14-18</sup>

70 The present research explores, for the first time, the effects of crystallization in a pharmaceutical  
71 polymeric system using DWS. Two eutectic systems with polyethylene glycol (PEG) and  
72 fenofibrate or flurbiprofen were employed as models.<sup>4,19</sup> This polymer type is hydrophilic with a  
73 crystalline lattice structure and can form eutectic mixtures with an active pharmaceutical  
74 ingredient (API). Specific interactions between PEG and a drug can suppress the crystalline  
75 polymer lattice to some degree during cooling of the molten blend. The model mixtures were  
76 selected for their ability to either specifically interact with PEG (e.g. flurbiprofen) or as a model  
77 for which no specific interactions are known (e.g. fenofibrate).<sup>4,20</sup> A particular aim of this DWS

78 pioneer study in polymer crystallization was to obtain insights into structure formation of the  
79 solidifying molten matrices upon cooling. This is not only of interest from a formulation  
80 perspective but also regarding manufacturing such as by hot melt extrusion.

81

## 82 **MATERIALS AND METHODS**

### 83 *Materials*

84 Fenofibrate was purchased from AK Scientific (30023 Ahern Ave Union City CA,USA), while  
85 poly(ethylene glycol) 6000, PEG was obtained from Sigma Aldrich (Riedstr. 2 D89559 Steinheim  
86 497329970). Flurbiprofen was supplied by Acros Organics (New Jersey USA) and uniform TiO<sub>2</sub>  
87 particles were obtained from LS Instruments (Fribourg, Switzerland).

88

### 89 *Methods*

#### 90 *Preparation of the eutectic mixtures*

91 Eutectic mixtures were prepared by using the melting method as described in literature.<sup>21</sup> Briefly,  
92 the PEG 6000 binary mixtures contained 24% (w/w) fenofibrate or alternatively 33% (w/w) of  
93 flurbiprofen. These concentrations were obtained from previous DSC measurements, (i.e. as  
94 mixtures with one endothermic event) as reported in literature.<sup>4</sup> The physical mixture was blended  
95 with a spatula in a metallic pan and heated up to 90°C in order to assure complete melting. The  
96 heat exposure during preparation and analysis of the eutectic mixtures was not expected to cause  
97 any drug degradation of the two rather stable compounds in line with previous reports that  
98 employed hot melt extrusion.<sup>22,23</sup> The obtained molten mixture in the present work was then cooled

99 down to room temperature and kept in a desiccator before analysis. Characteristics and  
100 compositions for various drug-PEG eutectic systems are described in **Table 1**.

101

### 102 *Powder x-ray diffraction (XRPD)*

103 Powder X-ray diffraction was used to characterize the solid form of the physical mixtures and of  
104 solid dispersions at ambient temperature using a Bruker D2 PHASER (Bruker AXS GmbH,  
105 Germany) with a PSD-50 M detector and EVA application software version 6. Samples were  
106 prepared by spreading powder samples on PMMA specimen holder rings from Bruker.  
107 Measurements were performed at 25°C with a Cu K $\alpha$  radiation source at 30 kV voltage, 10 mA  
108 current and were scanned from 6-40  $2\theta$ , with  $2\theta$  being the scattering angle at a scanning speed of  
109 2 °/min.

110

### 111 *Differential scanning calorimetry (DSC)*

112 A DSC 3 system (Mettler Toledo, Greifensee, Switzerland) was calibrated for temperature and  
113 enthalpy using indium. Nitrogen was used as the protective gas (200 mL/min). Samples  
114 (approximately 5 mg) were placed in 40  $\mu$ L aluminium pans with pierced aluminium lids. The  
115 melting point ( $T_m$ ) was determined by a single-segment heating ramp of 10 °C/min from 25 °C to  
116 a maximum temperature of 200 °C. All DSC measurements were carried out in triplicate.

117

### 118 *Fourier transform infrared spectroscopy*

119 Attenuated total reflection Fourier transform infrared (ATR-FTIR) spectra of pure compounds and

120 SDs were acquired in the 4000–600 cm<sup>-1</sup> range using a Cary 680 Series FTIR spectrometer  
121 (Agilent Technologies, Santa Clara, USA) equipped with an attenuated total reflectance accessory.  
122 A scanning range of 4000–600 cm was selected with 42 scans and a resolution of 4 cm. The spectra  
123 were evaluated using the software ACD/Spectrus Processor 2016.1.1 (Advanced Chemistry  
124 Development Toronto, Canada).

125

### 126 *Diffusing wave spectroscopy (DWS)*

127 **Measuring principle:** DWS is a light scattering technique that requires turbid samples to study the  
128 dynamic properties. In the transmission geometry, the sample is illuminated by a expanded laser  
129 light source and the transmitted light is analyzed on the opposite side.<sup>24</sup> The colloidal scattering  
130 particles can be present inside the sample, such for example oil droplets in emulsion, or they can  
131 be added in case of transparent sample such as, for example by dispersing titanium dioxide. Light  
132 detectors measure the intensity of the scattered light. The fluctuations of scattered light are  
133 characterized by the normalized intensity autocorrelation function (eq.1)<sup>24</sup>

$$134 \quad g^{(2)}(t) = \frac{\langle I(t_0)I(t_0+t) \rangle}{\langle I \rangle^2} \quad (1)$$

135

136 where the quantity  $\langle I \rangle$  is the average intensity, while  $t$  represents the lag time.

137 Using the Siegert relation (eq.2), the intensity correlation function and the field autocorrelation are  
138 related:

$$139 \quad g^{(2)}(t) = 1 + \beta |g^{(1)}(t)|^2 \quad (2)$$

140 where  $\beta$  is an instrumental factor given by the collection optics. Once the field correlation function  
 141 and  $l^*$  have been measured, the mean square displacement (MSD,  $\langle \Delta r^2(t) \rangle$ ) of a sample can be  
 142 calculated employing (eq.3)<sup>10,25</sup>:

143

$$144 \quad g^{(2)}(t)-1 \propto \left| \int_0^\infty P(s) \exp \left[ -\frac{1}{3} k^2 \langle \Delta r^2(t) \rangle \frac{s}{l^*} \right] ds \right|^2 \quad (3)$$

145

146 where  $k = 2\pi n/\lambda$  is the optical wavenumber including  $n$  as the refractive index of the medium and  
 147  $\lambda$  is the laser wavelength.  $P(s)$  represents the distribution of the photon trajectories of length  $s$  in  
 148 the sample of thickness  $L$ , while  $l^*$  is the transport mean free path which characterizes the typical  
 149 step length of the photon random walk.

150 The data obtained for the MSD were analyzed employing the function proposed by Bellour and  
 151 coworkers<sup>26</sup>:

$$152 \quad \langle \Delta r^2(t) \rangle = 6\delta^2 \left( 1 - e^{-\left(\frac{D_0 t}{\delta^2}\right)^\alpha} \right)^{1/\alpha} \left( 1 + \frac{D_m}{\delta^2} t \right) \quad (4)$$

153 Where  $\delta^2$  represents the amplitude of particle motion,  $t$  is the lag time,  $D_0$  and  $D_m$  are the short  
 154 and long time diffusion coefficient while  $\alpha$  is an additional parameter introduced to take into  
 155 account the broad spectrum of relaxation times at the plateau onset time.

156 Eq. 4 has been recently extended to better describe the region of longer relaxation times in case  
 157 of pharmaceutical emulsions.<sup>27</sup> On the other hand eq. 4 can also be simplified for other systems.  
 158 Bellour et al. suggested the following eq. 5 for particles that are harmonically bound (i.e.  
 159 “entrapped”) to exhibit Brownian motion around a stationary mean position:



160  $\langle \Delta r^2(t) \rangle = 6\delta^2 \left( 1 - e^{-\frac{D_0 t}{\delta^2}} \right)$  (5)

161 It is here possible to approximate the displacement value of the plateau with  $6\delta^2$ . The present  
162 work determined this plateau value based on the obtained MSD at the estimated inflection point.

163

164 ***Experimental setup:*** All samples were measured in transmission mode using a DWS RheoLab  
165 instrument (LS Instruments AG, Fribourg, Switzerland). The theory of DWS-based microrheology  
166 was already explained in detail in our previous work.<sup>16</sup> In brief, the laser light was scattered from  
167 the ground glass and collimated by a single lens before illuminating the sample. To avoid time  
168 consuming measurements at low frequencies, the instruments uses the so-called echo technique.  
169 The echo completes the data set to obtain the ICF over a broad range of lag times. In the echo  
170 mode the ground glass rotates during the measurement producing different illumination speckle  
171 pattern.<sup>15</sup> This feature is particularly important when working with non-ergodic samples, such as  
172 emulsions, gels and semi-crystalline polymers.<sup>15,16</sup> Samples were analyzed using a 5mm thickness  
173 cuvette. To ensure turbidity of the molten polymer and of the solid dispersion, 4.5 mg of titanium  
174 dioxide tracer particles with mean diameter of 360 nm were added and mixed with 2 g of sample.  
175 The mixture was poured into the cuvette to fill it up to 15-20 mm. The measurements were set in  
176 slow rotation mode at 300s with 30 s of echo mode. Five measurements of each temperature were  
177 done with two independent samples. The average of ten measurements is presented in all graphs.

178

### 179 ***Hot stage cross polarized microscopy***

180 An assessment of the crystal nucleation and growth was based on polarized light imaging using a  
181 heating stage coupled with a microscope Axioskop 2 mot (upright). The latter had a Hamamatsu

182 5810 3CCD video camera and was equipped with phase contrast, polarized filters, and DIC.  
183 Physical mixtures were molten at 85° degrees and the crystallization was studied using a remained  
184 constant magnification throughout the whole measurement (scale bars are displayed in every  
185 image).

186

187

188

### 189 *Molecular visualization by docking*

190 To qualitatively analyze and depict drug interactions with the polymer, molecular docking was  
191 conducted. Chemical structures were obtained from the ChemicaElectrica Gateway (v. 4.01) and  
192 loaded into Molecular Modeling Pro Plus (v.8.2.1.) (both programs by Norgwyn Montgomery  
193 Software Inc., North Wales, USA). A relatively shorter chain for polyethylene glycol (i.e. PEG  
194 400) was selected as model and this structure was drawn together with either the drug fenofibrate  
195 or flurbiprofen. Following a molecular mechanics minimization of the conformational energy  
196 (using the MM2 algorithm), partial charges of the molecules were calculated based on a semi-  
197 empirical quantum mechanical method (using a Complete Neglect of Differential Overlap, CNDO  
198 approach). Starting from 6 Angstroms distance between the molecular Van der Waals surfaces, a  
199 grid search algorithm proposed a molecular docking configuration of minimized energy. A relative  
200 permittivity of 18 was selected to approximate the PEG environment.<sup>28</sup>The final molecular  
201 association was depicted as combined wire frame and space filled model.

202

## 203 **RESULTS**

### 204 *Solid state characterizations*

205 Raw materials and the eutectic mixtures of PEG 6000 with either fenofibrate or flurbiprofen were  
206 characterized initially. Solid state analysis such as by DSC, PXRD and FT-IR was performed to  
207 determine the physical state of the raw materials and of the eutectic mixtures as well as to study  
208 interactions in both the molten and solid state.

209 While the composition of eutectic mixtures and the respective melting point are presented in **Table**  
210 **1**, the molecular docking of the two compounds with the carrier employed is presented in **Fig. 1**.

211 The DSC thermograms of pure PEG 6000, fenofibrate and flurbiprofen, as well as of their eutectic  
212 mixture are shown in **Fig. 2A**. The analysis of pure PEG 6000 shows that the onset of the  
213 endotherm occurs at 58 °C, having a peak at 62.5 °C. While the pure drug fenofibrate had a melting  
214 point of 80.2 °C and flurbiprofen's melting point was 114 °C, the endothermic event of the  
215 eutectics was much lower compared to the raw materials. The eutectic composition for PEG and  
216 fenofibrate has been determined previously to comprise 24% (w/w) of drug and it was independent  
217 of the molecular weight of the carrier.<sup>29</sup> Due to a strong hydrogen bonding, the eutectic  
218 composition for PEG and flurbiprofen was much higher with 33% (w/w) of API and strongly  
219 influenced by the molecular weight of the polymer.<sup>4,19</sup> As eutectic systems, the crystallinity of the  
220 pure drugs was affected in mixture regarding form as well as extent, thereby resulting in the  
221 observed DSC endotherms. The molecular interaction of the two model drugs with a PEG chain  
222 was also visualized based on molecular modeling (i.e. molecular docking of API to a polymer  
223 chain). **Fig. 1** depicts the hydrogen bond that is formed in the case of flurbiprofen whereas no such  
224 strong interaction was possible in case of fenofibrate with PEG.

225 The crystalline state of the raw materials and of the eutectics was analyzed further by PXRD. As  
226 observed in **Fig. 2B**, raw material and eutectics present distinct Bragg peaks, indicating a  
227 crystalline nature. When comparing the spectra of the raw flurbiprofen with that of the eutectic

228 mixture, a slight shift in the location of the peaks was observed. The interaction between  
229 flurbiprofen and PEG 6000 was further evaluated by FT-IR. Pronounced shifts in the absorption  
230 bands of FT-IR are visible in **Fig. 3** when comparing the spectra of flurbiprofen and fenofibrate as  
231 raw material and those obtained from the respective eutectic formulations. It can be seen that the  
232 carbonyl stretching band of flurbiprofen's carboxyl moiety at  $1700\text{ cm}^{-1}$  band has shifted to a  
233 higher frequency in the eutectic mixture. It has been argued that flurbiprofen is able to both donate  
234 and accept hydrogen bonds via the carboxyl moiety depending on the molecular weight of the  
235 polymer.<sup>4</sup> It is likely that flurbiprofen is able to donate hydrogen bonds with PEG 6000, while it  
236 might also accept hydrogen bonds with much lower molecular weight PEG considering a more  
237 relevant influence of hydroxyl groups present at the chain ends.<sup>30</sup> This is according to the expected  
238 molecular interaction based on molecular docking (**Fig. 1**) and the previously reported changes in  
239 FT-IR spectroscopy in mixtures of flurbiprofen and PEG.<sup>4,19</sup> Such a shift was not observed in the  
240 case of the fenofibrate and the eutectic mixture of this drug. There were also not pronounced  
241 changes in the lower wave number vibrations in case of the pure fenofibrate compared to that of  
242 the eutectic mixture. This was again different for the system of flurbiprofen where at relatively  
243 lower wave numbers, some changes were observed in the spectra; especially a peak at  $696.16\text{ cm}^{-1}$   
244 <sup>1</sup> that was shifted towards lower wavenumber ( $630.60\text{ cm}^{-1}$ ) for the solid dispersion of flurbiprofen  
245 with PEG.

## 246 *DWS*

247 Analysis of polymer crystallization and macromolecular structuring was a central aim of this study.  
248 Recent progress in DWS allows measuring at changing temperatures in a dynamic way so that a  
249 solidification of eutectic melts with and without drug could be analyzed.

250

251 **Fig. 4A** presents the intensity correlation function (ICF) of PEG 6000 from 85 °C to 48 °C. At 85  
252 °C when the polymer was completely molten, the intensity correlation function decayed to zero.  
253 Upon cooling, such decay occurred at longer lag time, and this abrupt shift was due to high  
254 viscosity increase that was expected by polymer nucleation and crystal growth since PEG 6000 is  
255 a semi-crystalline polymer at room temperature. The major changes in intensity correlation  
256 function were observed between 49 and 48 °C and the shift in lag time is displayed in **Fig. 4B**,  
257 where MSD is plotted. At relatively high temperature, the system was liquid and the MSD was  
258 linearly increasing with lag time, while with temperature decreased, the MSD was reaching a  
259 plateau. Such a plateau means that tracer particles exhibit limited Brownian motion as they become  
260 entrapped. Complex viscosity versus the frequency is presented in **Fig.4C**. While at high  
261 temperature, the polymer was behaving as a Newtonian system with constant viscosity across a  
262 broad range of frequencies, this was different at lower temperatures, for which viscosity was  
263 changing with frequency thereby suggesting a structuring in the course of polymer crystallization  
264 leading to shear thinning. Mean count rate (MCR) represents the average intensity of light  
265 (proportional to the number of photons arriving at the detector) that is an indicator of sample  
266 transparency. At the higher end of measured temperatures, MCR was around 300 kHz, whereas  
267 with the onset of polymer crystallization there was a strong decrease of the MCR, indicating an  
268 increase of turbidity. (**Fig. 4D**). This provided a sensitive analytical approach to the macroscopic  
269 appearance of the polymer that was transparent at 85°C but white at room temperature. In PEG  
270 6000 is a semi-crystalline polymer and the increase of turbidity was due to the crystallization of  
271 the polymer occurring around 48 °C.

272 The molten physical mixture of PEG 6000 and fenofibrate was then heated to 85 °C **Fig. 5A**  
273 compares the intensity correlation function (ICF) upon cooling to lower temperatures and 45.4°C

274 can be identified as the temperature of crystallization. Thus, at 85 °C when the polymer was  
275 completely molten, the ICF decayed to zero, whereas the ICF started to no longer reach zero upon  
276 cooling. ICFs were shifted to higher values of lag time(s) and the viscosity was increasing. The  
277 major changes in intensity correlation function for the eutectics were observed between 46 and  
278 45.4 °C. The shift in lag time is depicted in **Fig. 5B**, where MSD is plotted versus lag time. At  
279 high temperature for which the system is liquid, the MSD was increasing linearly with lag time,  
280 while upon decreasing the temperature, the MSD was reaching a plateau. Changes of the complex  
281 viscosity versus temperature are presented in **Fig.5C**. Similar to pure PEG 6000, the molten  
282 polymer mixture had a constant viscosity over the measured frequency range, whereas at lower  
283 temperatures, there was a decrease of viscosity over the frequency range suggesting again shear  
284 thinning as previously observed with pure PEG 6000. Further analysis of the fenofibrate and PEG  
285 6000 mixture is given by **Fig. 5D** that shows the decrease of the MCR over the temperature for  
286 this eutectic system. At higher temperatures, MCR was approximately 300 kHz and a strong  
287 decrease of the MCR, indicated an increase of turbidity in the course of matrix crystallization. The  
288 onset of crystallization was occurring at a temperature of about 45 °C and MCR reached  
289 approximately 220 kHz when the sample was completely solidified.

290 Finally, the model of the strongly interacting mixture was analyzed using PEG 6000 in  
291 combination with flurbiprofen. The molten mixture at 85 °C showed again an ICF that was  
292 decaying to zero. However, with a decrease of temperature there was again a shift to higher lag  
293 times, indicating a higher viscosity of the system as it can be seen in **Fig.6A**. For this eutectic  
294 mixture, no solidification was observed even close to the body temperature (38° C) in line with  
295 what has been reported before.<sup>4</sup> Again the MSD provided insights into the microstructure via the  
296 mean average distance that tracer particles were travelling. (**Fig. 6B**). In the liquid state, tracer

297 particle were able to perform Brownian motion and therefore, there was a linear increase of MSD  
298 versus lag time. For a nearly constant MSD, tracer particles were apparently confined around their  
299 mean positions. When considering **Fig 6C**, the systems does not exhibit an ideally viscous  
300 behavior at high temperature; therefore, an apparent shear thickening may have caused the  
301 observed frequency changes at 85°C, while cooling of the system was leading to a rheological  
302 behavior suggesting shear thinning similar to the previously analyzed samples. Such cooling led  
303 again to a strong decrease of MCR values due to the occurrence of crystallites as shown **Fig 6D**.  
304 Since all samples showed a plateau regime of MSD, a comparison for this apparent confinement  
305 of the tracer particles due to microstructuring of the solidifying polymer matrix. Following the  
306 approach of Bellour et al.<sup>26</sup> (Eq. 5) for the present case of polymeric melts, the given confinement  
307 of the tracer particles can be plotted as a kind of “cage size” as displayed in **Fig. 7**. The dimension  
308 of this apparent confinement of tracer particles was biggest for polymer alone, followed by the  
309 eutectic mixtures with fenofibrate and finally that with flurbiprofen. The data of DWS were finally  
310 complemented by studies using hot stage cross polarized microscopy and the results are presented  
311 in **Fig.8A, B and C** respectively. As mentioned before, polymer drug interaction play a significant  
312 role in polymer chain folding during the crystallization and as highlighted in the **Fig. 8C**, the  
313 crystalline structure of the SD with flurbiprofen in the solid state appeared to be different compared  
314 to the polymer alone or also compared to the SD with fenofibrate that was less interacting with the  
315 polymer matrix. It is evident that the crystalline structure was in the latter case rather disrupted in  
316 line with previous reports in the literature.<sup>5,19,29</sup>

317

## 318 **DISCUSSION**

319 Crystallization of polymers is a complex topic and the classical of Hoffmann-Lauritzen theory  
320 provides a basic understanding of how polymeric chains exhibit surface crystallization into  
321 lamellae of a given thickness.<sup>31</sup> Since these pioneer days, several more refined theoretical models  
322 involving multistep crystallization have been proposed and a very recent review in the journal  
323 *Macromolecules* comes to the conclusion that even these days, important theoretical questions  
324 remain unanswered.<sup>32</sup> The theoretical complexity is of course even increased in presence of  
325 additives such as a drug and therefore, pharmaceutical research has mostly taken just a  
326 phenomenological approach to crystallization of polymeric drug formulations. However, any  
327 experimental study of a phase transition such as crystallization is also difficult because the  
328 analytics should not disturb the observed process, which is a concern with techniques such as  
329 mechanical rheology. The present work therefore used for the first time broadband DWS to study  
330 pharmaceutical eutectic model systems in the course of solidification upon cooling.

331  
332 Previous work suggested that flurbiprofen is strongly interacting with PEG while fenofibrate is  
333 not.<sup>4</sup> Solid state characterization employing DSC, PXRD were performed and the results are shown  
334 in **Fig. 2**. Specific interactions such as hydrogen bonding between flurbiprofen and PEG 6000 have  
335 been studied with FT-IR and presented in **Fig 3**. The current study found that drug-polymer  
336 interaction played a key role in case of flurbiprofen during phase transition of the PEG-based solid  
337 dispersion. It can be well imagined that strongly attached flurbiprofen would affect polymer chain  
338 flexibility and its bulkiness so that crystalline packing of lamellae would be affected. This  
339 qualitative view may explain the observed effects on polymer matrix crystallization in case of this  
340 strongly interacting eutectic system. The solid dispersion comprising PEG 6000 and fenofibrate  
341 exhibited ideal viscous behavior at high temperatures and the frequency spectrum suggested some



342 viscoelastic behavior at lower temperatures. (Fig. 4C and 5C). The high temperature regime can  
343 be imagined to hold for a typical melt with polymer chains that can arrange in random coils and  
344 thereby provide a rather homogenous system with practically ideal viscous behavior. Once  
345 polymer nucleation and growth occurs, the lamellar grow into crystallites and such spherulites can  
346 further aggregate. Since these aggregates provide suspended particles, there was turbidity noted as  
347 indicated by the MCR of the DWS experiments. The frequency dependence of viscosity was  
348 analogues to shear thinning of a suspension and a recent study actually evaluated typical models  
349 of suspension rheology for such samples.<sup>33</sup>The authors pointed to the analogy of particle hierarchy  
350 between aggregated crystallites and other polymeric nanocomposites. The latter nanocomposites  
351 are typically agglomerates of small aggregates and these again consist of primary nanoparticles.  
352 Analogously, crystal aggregates (spherulites) are aggregates of crystal lamellae consisting of  
353 several individual lamellae.

354

355 Aggregated polymeric crystals can exhibit mechanical rigidity and therefore elastic behavior.  
356 Determination of the solidification point is obtained using the crossover point of  $G'$  and  $G''$  and  
357 the results presented in **Table 2**. Indeed, above the given crystallization temperature, samples were  
358 in molten state exhibiting  $G'' > G'$ , while upon solidification, there was dominance of the elastic  
359 modulus with  $G' > G''$ . (data not shown)

360 Newtonian behavior at high temperatures was not found in case of the molten solid dispersion with  
361 flurbiprofen. (Fig. 6C) The changes along the differences frequencies suggested increasing  
362 viscosity at 85 °C, whereas upon cooling, data indicated again a shear thinning behavior. The latter  
363 behavior supports the view that occurrence of crystallites was leading to a suspension-type of  
364 rheology. However, the behavior at 85°C in the melt is particularly notable since PEG 6000 alone

365 did not show such increased viscosity with frequency but was rather Newtonian. Polymers of  
366 higher molecular weight can show an increase of viscosity with rising frequency which is due to  
367 the entanglement of longer chains. Each detanglement is expected to require a specific relaxation  
368 time and for higher frequencies, transient bonds of entanglement would become permanent on the  
369 given time scale.<sup>34</sup> Even though a strongly interacting small-molecular drug like flurbiprofen  
370 would not greatly lengthen the chains of PEG 6000, the present results still display some frequency  
371 dependency of the viscous melt caused by the presence of flurbiprofen.

372 The presence of strong molecular interactions between a drug and PEG 6000 was also studied by  
373 Van Duong et al. (2017). They outlined the view that in the melt, PEG chains are locked in  
374 hydrogen bonding with drug molecules and therefore no more than one repeated unit of helical  
375 structure can be folded. The crystallization process for the considered systems is taking place in a  
376 relatively narrow temperature range, and this may occur especially when the system is  
377 homogeneously dispersed. The model assumes that in the course of chain folding, drug-polymer  
378 hydrogen bonds are disrupted leading to a segregation of API. Some drug remains hydrogen  
379 bonded to the surface of the folded lamellae and is part of drug-rich domains. Such solidified  
380 polymer systems are typically semi-crystalline and the extent of crystallinity as well as the given  
381 microstructure were previously mentioned to likely affect quality attributes such as drug release  
382 .<sup>4,5</sup>

383 To gain a better understanding of how the presence of drug affected polymer crystallization, the  
384 average movement of the tracer particles as MSD grants insights into the microstructuring in the  
385 course of crystallization. The approach by Bellour et al.<sup>26</sup> has been previously used to describe  
386 other structured liquids such as a surfactant solution of worm-like micelles and it was qualitatively  
387 used in the present study to interpret the obtained data of the polymeric melts upon

388 crystallization.<sup>26</sup> When tracer particles exhibit a nearly constant MSD for a regime of frequencies  
389 (or lag times), then there is a kind of apparent entrapment given. The so-called “cage size” grants  
390 indirect insights into the structure of the matrix that is surrounding the used tracer particles. It is  
391 noteworthy that the MSD decreases in presence of polymer drug interaction, which is also  
392 reflecting by the apparent cage size at the inflexion point (named as  $6\delta^2$ ) as presented in **Fig.7**.  
393 Given that tracer particles were embedded in a matrix of crystallizing and aggregating lamellae,  
394 the differences in apparent cage size may suggest how finely meshed these networks of lamellae  
395 were. This would support the view that the strongly interacting flurbiprofen perturbs polymer  
396 crystallization thereby leading to relatively smaller lamellae compared with pure PEG 6000 or its  
397 SD with fenofibrate. Indeed, the presence of flurbiprofen attached to the PEG via hydrogen  
398 bonding would inhibit the crystallization of the polymer, because it induces defects in the PEG  
399 crystalline network, hindering it from growing and structuring in line with results depicted in **Fig.8**.  
400 The present findings show that microstructuring of polymer in presence of drug can be studied not  
401 only by small-angle X scattering but also with novel microrheological tool such as broadband  
402 DWS.<sup>35,36</sup>

403 These findings help us to understand the microstructuring during phase transition of the polymer  
404 and of eutectic mixtures at high frequencies and in a non-invasive conditions. In addition,  
405 physicochemical properties and pharmaceutical performance of PEG-based solid dispersion  
406 depend on the drug-polymer interactions. Disruption of the crystalline lattice has also expected  
407 implications of drug release in that a relatively lower matrix crystallinity typically shows faster  
408 drug release compared to an eutectic system with a higher degree of crystallization. Therefore, the  
409 given microstructure is central for the quality attributes of the given SD formulations and present

410 work may find applications in formulation development as well as in process development of a  
411 eutectic drug product.

412

## 413 **CONCLUSIONS**

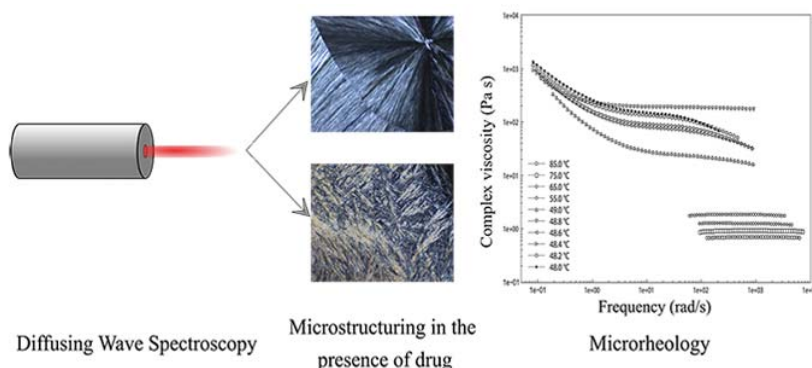
414 The physico-chemical properties of polymeric drug carriers and of the final dosage form depend  
415 on the microstructuring during the crystallization process in case of polymer eutectics. DWS and  
416 MSD were employed to study macromolecular structuring during cooling of the PEG-based  
417 mixtures with two model drugs, fenofibrate and flurbiprofen. While the first compound was barely  
418 interacting with the polymer, flurbiprofen provided a distinct molecular interaction with the  
419 carrier. This interaction already present in the molten state was changing the rheological behavior  
420 of the otherwise pure polymer melt. The crystallization of polymer was monitored by studying  
421 complex viscosity and MSD parameters from DWS. Indirectly it was possible to gain insights into  
422 how lamellae may crystalize and aggregate in the different formulations. Given the importance of  
423 the microstructure of such eutectic systems on different pharmaceutical quality attributes, the  
424 present findings are of high relevance also for practical formulation development. Compared to  
425 classical rheological measurements where the directions of the crystallizing lamellae are  
426 influenced during the measurements, DWS offers insights into the microstructuring of crystallizing  
427 lamellae based on contact-free measurements over a broad frequency range. Such mechanistic  
428 analysis and understanding of microstructuring under non-invasive conditions is not only relevant  
429 for eutectic systems. Other solid drug dispersions could be studied in the future too and tracer  
430 particles should be added whenever a system does not provide sufficient light scattering on its  
431 own. Therefore, DWS is a quite versatile tool to study the solidification behavior of drug-excipient

432 mixtures, which is important to properly understand phase behavior and microstructuring of  
433 pharmaceutical formulations.

434

435

436 **TOC**



437

438 This paper employs Diffusing Wave Spectroscopy (DWS) over a broad frequency band to study  
439 polymer-drug systems in a non-invasive way. Eutectic mixtures of polyethylene glycol (PEG) were  
440 investigated using two model drugs: fenofibrate and flurbiprofen. While the first model compound  
441 was barely interacting with the polymer, flurbiprofen showed pronounced molecular interaction  
442 with the polymer, thereby influencing the microstructuring of the system.

443

444

445

446

447 **FIGURES**

448 **Fig. 1.** Molecular docking of PEG 6000 with fenofibrate (A) and flurbiprofen (B)

449 **Fig. 2.** Differential Scanning Calorimetry (DSC) (A) and Powder X-ray diffraction (XRPD) (B)  
450 plots of fenofibrate (a), flurbiprofen (b), PEG c) PEG –fenofibrate SD (d) and PEG- flurbiprofen  
451 SD (e).

452 **Fig. 3.** FT-IR of flurbiprofen (gray solid line), solid dispersion with flurbiprofen (gray dots),  
453 fenofibrate (black solid line), solid dispersion with fenofibrate (black dots).

454 **Fig. 4.** Microrheological characterization of PEG 6000: ICF (A), MSD (B), complex viscosity (C)  
455 and MCR (D).

456 **Fig. 5.** Microrheological characterization of solid dispersion of PEG 6000 and fenofibrate: ICF  
457 (A), MSD (B), complex viscosity (C) and MCR (D).

458 **Fig. 6.** Microrheological characterization of solid dispersion of PEG 6000 and flurbiprofen: ICF  
459 (A), MSD (B), complex viscosity (C) and MCR (D).

460 **Fig. 7.** Comparison of the apparent cage size (named as  $6 \square 2$ ) in pure PEG, SD of fenofibrate and  
461 SD of flurbiprofen close to their solidification temperature.

462 **Fig. 8.** Hot stage cross polarized light microscopy of PEG 6000 (A), SD of fenofibrate (B) and SD  
463 of flurbiprofen (C)

464

465 **TABLES**466 **Table 1.** Characteristics and composition for various Drug-PEG eutectic systems

467

Compound	Molecular weight (MW)	$T_m$ (°C)	$\Delta H_f$ (kJ/mol e)	Molar volume (cm <sup>3</sup> /mol)	cLogP (n-octanol/water)	Experimental eutectic composition with PEG 6000 (%)	References
Fenofibrate	361	80.2	34.0	310.7	4.43	24	20
Flurbiprofen	244	114.7	28.0	263.5	4.12	33	19
PEG	6000	58-63	-	-	-	-	

468

469

470

471 **Table 2.** Temperatures and  $G'$   $G''$  values at the intersection of curves

	Temperature (°C)	Frequency (rad/s)	Moduli $G'$ $G''$ (Pa)
PEG	48.20	0.63	120.00
	48.00	0.70	150.60
SD of fenofibrate	45.60	2.69	19.54
	45.40	0.50	398.58
SD of flurbiprofen	38.40	0.12	151.40
	38.20	1.64	1958.76

472

473

474

475 **AUTHOR INFORMATION**

476 **Corresponding Author**

477 **Martin Kuentz** - *University of Applied Sciences and Arts Northwestern Switzerland, Institute of*  
478 *Pharma Technology, Muttenz, CH-4132 Switzerland, E-mail: [martin.kuentz@fhnw.ch](mailto:martin.kuentz@fhnw.ch)*

479

480 **Authors**

481 **Sandra Jankovic** - *University of Applied Sciences and Arts Northwestern Switzerland, Institute*  
482 *of Pharma Technology, Muttenz, CH-4132 Switzerland; University of Basel, Department of*  
483 *Pharmaceutical Sciences, Basel, Switzerland*

484

485 **Simone Aleandri** - *University of Bern, Department of Chemistry and Biochemistry, Bern,*  
486 *Switzerland*

487 **Mathias Reufer** - *Zumbach Electronic, Hauptstrasse 93, Postfach, 2552 Orpund, Switzerland*

488

489 **ACKNOWLEDGMENT**

490 This project has received funding from the European Union's Horizon 2020 Research and  
491 Innovation Programme under grant agreement No 674909 (PEARRL). We are grateful to Prof. Dr.  
492 Mezzenga for the use of the hot stage cross polarized microscopy at ETH and Theodor Bühler for  
493 the assistant in FT-IR measurements. We further thank Prof. Frank Scheffold for his valuable  
494 comments in the preparation of the manuscript.

495



496 **REFERENCES**

- 497 (1) Matsen, M. W. *Soft Matter*, Volume 1: Polymer Melts and Mixtures. Wiley-VCH  
498 Weinheim 2006.
- 499 (2) Singh, Y. *Martin's Physical Pharmacy and Pharmaceutical Sciences. Rutgers, State Univ.*  
500 *New Jersey* **2006**.
- 501 (3) Martin, A.; Swarbrick, J.; Cammarata, A. *Physical Pharmacy. Lea Febiger* **2006**.
- 502 (4) Vippagunta, S. R.; Wang, Z.; Hornung, S.; Krill, S. L. Factors Affecting the Formation of  
503 Eutectic Solid Dispersions and Their Dissolution Behavior. *J. Pharm. Sci.* **2007**, *96* (2),  
504 294–304. <https://doi.org/10.1002/jps.20754>.
- 505 (5) Van Duong, T.; Reekmans, G.; Venkatesham, A.; Van Aerschot, A.; Adriaensens, P.; Van  
506 Humbeeck, J.; Van den Mooter, G. Spectroscopic Investigation of the Formation and  
507 Disruption of Hydrogen Bonds in Pharmaceutical Semicrystalline Dispersions. *Mol. Pharm.*  
508 **2017**, *14* (5), 1726–1741. <https://doi.org/10.1021/acs.molpharmaceut.6b01172>.
- 509 (6) Broman, E.; Khoo, C.; Taylor, L. S. A Comparison of Alternative Polymer Excipients and  
510 Processing Methods for Making Solid Dispersions of a Poorly Water Soluble Drug. *Int. J.*  
511 *Pharm.* **2001**, *222* (1), 139–151. [https://doi.org/10.1016/S0378-5173\(01\)00709-8](https://doi.org/10.1016/S0378-5173(01)00709-8).
- 512 (7) De Kee, D.; Wissbrun, K. F. Polymer Rheology. *Phys. Today* **1998**, *51* (6), 24–29.  
513 <https://doi.org/10.1063/1.882283>.
- 514 (8) Röntzsch, V.; Özen, M. B.; Rätzsch, K.; Stellamanns, E.; Sprung, M.; Guthausen, G.;  
515 Wilhelm, M. Polymer Crystallization Studied by Hyphenated Rheology Techniques: Rheo-  
516 NMR, Rheo-SAXS, and Rheo-Microscopy. *Macromol. Mater. Eng.* **2019**, *304* (2),  
517 1800586. <https://doi.org/10.1002/mame.201800586>.
- 518 (9) Del Giudice, F.; Tassieri, M.; Oelschlaeger, C.; Shen, A. Q. When Microrheology, Bulk  
519 Rheology, and Microfluidics Meet: Broadband Rheology of Hydroxyethyl Cellulose Water  
520 Solutions. *Macromolecules* **2017**, *50* (7), 2951–2963.  
521 <https://doi.org/10.1021/acs.macromol.6b02727>.
- 522 (10) Dasgupta, B. R.; Tee, S.-Y.; Crocker, J. C.; Frisken, B. J.; Weitz, D. A. Microrheology of  
523 Polyethylene Oxide Using Diffusing Wave Spectroscopy and Single Scattering. *Phys. Rev.*  
524 *E* **2002**, *65* (5), 051505. <https://doi.org/10.1103/PhysRevE.65.051505>.
- 525 (11) Pine, D. J.; Weitz, D. A.; Zhu, J. X.; Herbolzheimer, E. Diffusing-Wave Spectroscopy:  
526 Dynamic Light Scattering in the Multiple Scattering Limit. *J. Phys.* **1990**, *51* (18), 2101–  
527 2127. <https://doi.org/10.1051/jphys:0199000510180210100>.
- 528 (12) MacKintosh, F. C.; John, S. Diffusing-Wave Spectroscopy and Multiple Scattering of Light  
529 in Correlated Random Media. *Phys. Rev. B* **1989**, *40* (4), 2383–2406.  
530 <https://doi.org/10.1103/PhysRevB.40.2383>.
- 531 (13) Pine, D. J.; Weitz, D. A.; Chaikin, P. M.; Herbolzheimer, E. Diffusing Wave Spectroscopy.  
532 *Phys. Rev. Lett.* **1988**, *60* (12), 1134–1137. <https://doi.org/10.1103/PhysRevLett.60.1134>.

- 533 (14) Jankovic, S.; O'Dwyer, P. J.; Box, K. J.; Imanidis, G.; Reppas, C.; Kuentz, M. Biphasic  
534 Drug Release Testing Coupled with Diffusing Wave Spectroscopy for Mechanistic  
535 Understanding of Solid Dispersion Performance. *Eur. J. Pharm. Sci.* **2019**, *137*, 105001.  
536 <https://doi.org/10.1016/j.ejps.2019.105001>.
- 537 (15) Reufer, M.; Machado, A. H. E.; Niederquell, A.; Bohnenblust, K.; Müller, B.; Völker, A.  
538 C.; Kuentz, M. Introducing Diffusing Wave Spectroscopy as a Process Analytical Tool for  
539 Pharmaceutical Emulsion Manufacturing. *J. Pharm. Sci.* **2014**, *103* (12), 3902–3913.  
540 <https://doi.org/10.1002/jps.24197>.
- 541 (16) Niederquell, A.; Völker, A. C.; Kuentz, M. Introduction of Diffusing Wave Spectroscopy  
542 to Study Self-Emulsifying Drug Delivery Systems with Respect to Liquid Filling of  
543 Capsules. *Int. J. Pharm.* **2012**, *426* (1–2), 144–152.  
544 <https://doi.org/10.1016/j.ijpharm.2012.01.042>.
- 545 (17) Alexander M, Piska I, D. D. Investigation of Particle Dynamics in Gels Involving Casein  
546 Micelles: A Diffusing Wave Spectroscopy and Rheology Approach. *Food Hydrocoll.* **2008**,  
547 *22* (6), 1124–1134. <https://doi.org/10.1016/J.FOODHYD.2007.06.004>.
- 548 (18) Aleandri, S.; Jankovic, S.; Kuentz, M. Towards a Better Understanding of Solid Dispersions  
549 in Aqueous Environment by a Fluorescence Quenching Approach. *Int. J. Pharm.* **2018**, *550*  
550 (1–2), 130–139. <https://doi.org/10.1016/j.ijpharm.2018.08.029>.
- 551 (19) Lacoulonche, F.; Chauvet, A.; Masse, J. An Investigation of Flurbiprofen Polymorphism by  
552 Thermoanalytical and Spectroscopic Methods and a Study of Its Interactions with Poly-  
553 (Ethylene Glycol) 6000 by Differential Scanning Calorimetry and Modelling. *Int. J. Pharm.*  
554 **1997**, *153* (2), 167–179. [https://doi.org/10.1016/S0378-5173\(97\)00102-6](https://doi.org/10.1016/S0378-5173(97)00102-6).
- 555 (20) Law, D.; Wang, W.; Schmitt, E. A.; Qiu, Y.; Krill, S. L.; Fort, J. J. Properties of Rapidly  
556 Dissolving Eutectic Mixtures of Poly(Ethylene Glycol) and Fenofibrate: The Eutectic  
557 Microstructure. *J. Pharm. Sci.* **2003**, *92* (3), 505–515. <https://doi.org/10.1002/jps.10324>.
- 558 (21) Chiou, W. L.; Riegelman, S. Pharmaceutical Applications of Solid Dispersion Systems. *J.*  
559 *Pharm. Sci.* **1971**, *60* (9), 1281–1302. <https://doi.org/10.1002/jps.2600600902>.
- 560 (22) Ditzinger, F.; Scherer, U.; Schönenberger, M.; Holm, R.; Kuentz, M. Modified Polymer  
561 Matrix in Pharmaceutical Hot Melt Extrusion by Molecular Interactions with a Carboxylic  
562 Cofomer. *Mol. Pharm.* **2018**, *16* (1), 141–150.
- 563 (23) Cheng, L.; Li, T.; Dong, L.; Wang, X.; Huo, Q.; Wang, H.; Jiang, Z.; Shan, X.; Pan, W.;  
564 Yang, X. Design and Evaluation of Bilayer Pump Tablet of Flurbiprofen Solid Dispersion  
565 for Zero-Order Controlled Delivery. *J. Pharm. Sci.* **2018**, *107* (5), 1434–1442.
- 566 (24) Furst, E. M.; Squires, T. M. *Microrheology*, 1st ed.; OUP Oxford, Ed.; OUP Oxford, 2017.
- 567 (25) Constantin, D.; Knaebel, A.; Bellour, M.; Padding, J. T.; Boek, E. S. *Microrheology of*  
568 *Giant-Micelle Solutions.* **2002**.
- 569 (26) Bellour, M.; Skouri, M.; Munch, J.-P.; Hébraud, P. Brownian Motion of Particles Embedded  
570 in a Solution of Giant Micelles. *Eur. Phys. J. E* **2002**, *8* (4), 431–436.

- 571 <https://doi.org/10.1140/epje/i2002-10026-0>.
- 572 (27) Niederquell, A.; Machado, A. H. E. E.; Kuentz, M. A Diffusing Wave Spectroscopy Study  
573 of Pharmaceutical Emulsions for Physical Stability Assessment. *Int. J. Pharm.* **2017**, *530*  
574 (1–2), 213–223. <https://doi.org/10.1016/j.ijpharm.2017.07.038>.
- 575 (28) Niederquell, A.; Dujovny, G.; Probst, S. E.; Kuentz, M. A Relative Permittivity Approach  
576 for Fast Drug Solubility Screening of Solvents and Excipients in Lipid-Based Delivery. *J.*  
577 *Pharm. Sci.* **2019**, *108* (10), 3457–3460. <https://doi.org/10.1016/j.xphs.2019.06.014>.
- 578 (29) Law, S. L.; Lo, W. Y.; Lin, F. M.; Chaing, C. H. Dissolution and Absorption of Nifedipine  
579 in Polyethylene Glycol Solid Dispersion Containing Phosphatidylcholine. *Int. J. Pharm.*  
580 **1992**, *84* (2), 161–166. [https://doi.org/10.1016/0378-5173\(92\)90056-8](https://doi.org/10.1016/0378-5173(92)90056-8).
- 581 (30) Lacoulonche, F.; Chauvet, A.; Masse, J.; Egea, M. A.; Garcia, M. L. An Investigation of  
582 FB Interactions with Poly(Ethylene Glycol) 6000, Poly(Ethylene Glycol) 4000, and Poly-  
583  $\epsilon$ -Caprolactone by Thermoanalytical and Spectroscopic Methods and Modeling. *J. Pharm.*  
584 *Sci.* **1998**, *87* (5), 543–551. <https://doi.org/10.1021/js970443+>.
- 585 (31) Lauritzen, J. I.; Hoffman, J. D. Formation of Polymer Crystals with Folded Chains from  
586 Dilute Solution. *J. Chem. Phys.* **1959**, *31* (6), 1680–1681.  
587 <https://doi.org/10.1063/1.1730678>.
- 588 (32) Tang, X.; Chen, W.; Li, L. The Tough Journey of Polymer Crystallization: Battling with  
589 Chain Flexibility and Connectivity. *Macromolecules* **2019**, *52* (10), 3575–3591.  
590 <https://doi.org/10.1021/acs.macromol.8b02725>.
- 591 (33) He, P.; Yu, W.; Zhou, C. Agglomeration of Crystals during Crystallization of  
592 Semicrystalline Polymers: A Suspension-Based Rheological Study. *Macromolecules* **2019**,  
593 *52* (3), 1042–1054. <https://doi.org/10.1021/acs.macromol.8b02452>.
- 594 (34) Ballard, M. J.; Buscall, R.; Waite, F. A. The Theory of Shear-Thickening Polymer  
595 Solutions. *Polymer (Guildf)*. **1988**, *29* (7), 1287–1293. [https://doi.org/10.1016/0032-3861\(88\)90058-4](https://doi.org/10.1016/0032-3861(88)90058-4).
- 597 (35) MacMillan\*, S. D.; Roberts, K. J.; Rossi, A.; And, M. A. W.; Polgreen, M. C.; Smith, I. H.  
598 In Situ Small Angle X-Ray Scattering (SAXS) Studies of Polymorphism with the  
599 Associated Crystallization of Cocoa Butter Fat Using Shearing Conditions. **2002**.  
600 <https://doi.org/10.1021/CG0155649>.
- 601 (36) Strobl, G. R.; IUCr. Determination of the Lamellar Structure of Partially Crystalline  
602 Polymers by Direct Analysis of Their Small-Angle X-Ray Scattering Curves. *J. Appl.*  
603 *Crystallogr.* **1973**, *6* (5), 365–370. <https://doi.org/10.1107/S0021889873008897>.

604

605